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Ultra Sensitive Micro String Resonators for Solid State Thermo-mechanical Analysis of Small and Large Molecules

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KEYWORDS. thermal analysis, MEMS, resonator, string, β-relaxation, glass transition

ABSTRACT: Thermal analysis plays an important role in both industrial and fundamental research and is widely used to study thermal characteristics of a variety of materials. However, despite considerable effort using different techniques, research struggles to resolve the physico-chemical nature of many thermal transitions such as amorphous relaxations or structural changes in proteins. To overcome the limitations in sensitivity of conventional techniques and to gain new insight into the thermal and mechanical properties of small and large molecule samples, we have developed an instrumental analysis technique using resonating low stress silicon nitride microstrings. With a simple sample deposition method and post process data analysis, we are able to perform rapid thermal analysis of direct instrumental triplicate samples with only pico- to nanograms of material. Utilizing this method, we present the first measurement of amorphous alpha and beta relaxation, as well as liquid crystalline transitions and decomposition of small molecule samples deposited onto a micro string resonator. Furthermore, sensitive measurements of the glass transition of polymers and yet unresolved thermal responses of proteins below their apparent denaturation temperature, which seem to include the true solid state glass transition of pure protein, are reported. Where applicable, thermal events detected with the setup were in good agreement with conventional techniques such as differential scanning calorimetry and dynamic mechanical analysis. The sensitive detection of even subtle thermal transitions highlights further possibilities and applications of resonating microstrings in instrumental physico-chemical analysis.

INTRODUCTION

Thermal analysis comprises of an important analytical tool-box for the physico-chemical characterization of a variety of analytes across many fields of chemical sciences: for material and polymer scientists, first and second order phase transitions are crucial for material processability as well as for the application range of the final product. Pharmaceutical research, often focusing on small molecules or proteins as active ingredients, is particularly interested in transitions related to the chemical and physical stability of the formulation. A range of methods has been established to analyze these materials as they change with temperature, each with its own advantages and limitations. Here, we introduce an instrumental analysis technique using resonating silicon nitride (Si,N) microstring sensors to study the thermal and mechanical properties of small molecule drugs, polymers and proteins with only pico- to nanograms of sample.

Resonating Micro- and Nano Electro Mechanical Systems (MEMS, NEMS) of different geometries have been utilized for a variety of applications such as oscillators and filters, as well as chemical, mass and humidity sensors.1-5 However, research into probing thermal characteristics of samples deposited onto the sensors is still in its infancy with only a few proof-of-concept studies reported in the literature.6-14 All but one of these studies focus on polymers. Furthermore, the main limitations are the applied sample deposition methods: specialized spray- or spin coating, ink jet printing or melt infiltration are used to deposit the sample on the sensors. Spray coating and ink jet printing can only be applied to specific samples and their process optimization is tedious.15-26 Melt infiltration combined with a specific suspended microchannel geometry can only be used if the sample does not degrade upon heating or melting and if the sample possesses a sufficiently low melt viscosity.7 Importantly, all of these methods physically alter the sample.

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The sampling technique used in this study is a direct solid state deposition method (Figure 1a). The sample is transferred onto a sieve, which is used for initial size reduction and placed above the sampling chamber. A vacuum is applied through a hole, drilled in the bottom part of the chamber. The resulting air flow forces the samples to deposit on the strings (Figure 1b) by inertial impaction. Furthermore, a shadow mask is used to define the sample location on the string sensor. After the sampling procedure, the sensor chip is transferred to a homemade temperature stage, placed in a vacuum chamber. A laser-Doppler-vibrometer is used for readout of the string sensor and a controller sets and monitors the temperature profiles. An overview of the measurement setup is given in Figure 1c.

The measuring principle used in this work is based on tracking changes in the resonant behavior of the microstring sensors and relating them to changes in the analyzed sample. Every mechanical structure has a fundamental mechanical Eigenfrequency that is a function of its material properties and geometry. In real world structures where mechanical losses are present, these frequencies are called resonance frequencies. As seen by equation (1), an increase in stiffness or tensile stress or a decrease in effective mass lead to higher resonance frequencies of string resonators.

$$f_{\text{res},n} = \frac{n}{2l} \sqrt{\frac{\sigma}{\rho}}, n = 1,2,3...$$  (1)

n: resonance mode number, l: length of the string
\(\sigma\): tensile stress, \(\rho\): mass density

Determining the quality factor (Q), a measure of the dissipation of mechanical energy in the resonator system, reveals additional information about the viscous damping in the sample.

During the experiments, the entire microchip is heated. As the silicon frame expands more than the SiN strings during heating, the strings’ tensile stress increases, leading to higher resonance frequencies. Thermal transitions of samples deposited onto the string affect the physical properties of the system and result in frequency slope changes, as well as in changes of Q.

The method presented here is referred to as Micromechanical Thermal Analysis (MTA) and has many advantages compared to conventional techniques. It allows the use of very small sample sizes (0.1 to 345 ng), which is beneficial in settings where only small amounts of compounds are available, such as in pharmaceutical research and development. With the setup used in this study, heating rates of 40°C/min can be achieved, while conventional techniques such as Modulated Temperature Differential Scanning Calorimetry (mDSC) and Dynamic Mechanical Analysis (DMA) are usually limited to heating rates of 0.5 – 5°C/min.

The measurement laser is able to jump between individual strings clamped onto the same chip. In this way, spectra for samples on every string can be recorded and \(f_0\) and Q are extracted from the frequency response spectra after the measurement has finished. As sampling through inertial impaction is an inherently random process, each string supplies an independent measurement. In our study, three strings per microchip were measured and analyzed simultaneously, to create an instant instrumental triplicate. These advances drastically decrease the measurement time per sample.

Using this method, we hereby show the first measurements of both the alpha and the beta relaxation of amorphous small molecules, as well as liquid crystalline phase transitions probed with a micromechanical string resonator. These processes are of great interest in pharmaceutical research, as amorphous formulations are a promising approach to counteract the negative impact of low aqueous solubility on bioavailability. The use of amorphous formulations is however limited to their unstable nature, leading to only a few marketed products. The alpha re-
laxation (or glass transition, $T_g$) has previously been described as an important indicator for amorphous stability.\textsuperscript{21} Moreover, recent focus of physico-chemical investigation of glassy materials is also including the beta relaxation $^{15}$ ($T_{\beta}$), which is only accessible by more sensitive measuring techniques than mDSC, that require bulk sample sizes ($> 100$ mg) such as Dielectric Spectroscopy\textsuperscript{26} and DMA\textsuperscript{23} or more advanced vibrational spectroscopic techniques such as Terahertz Spectroscopy.\textsuperscript{26}

In this study, we also present sensitive measurements of the $T_g$ of polymers. Besides its importance for amorphous stability and processability, the $T_g$ is an important parameter for the application range of the polymer. For example, rubber elastomers are used above their $T_g$, whereas amorphous plastics are mostly used below their $T_g$.\textsuperscript{27}

The solid state sampling technique used in this study, also allows for the probing of protein samples. An important thermal transition of proteins with respect to stability as well as biological activity is the denaturation onset. Moreover, pharmaceutical research and technology, as well as the field of biomimetics are interested in thermal phenomena that happen below the denaturation temperature for reasons of stability, processability and biological activity.\textsuperscript{28-29} Therefore it is important to have sensitive tools to better characterize these samples. Consequently, this study includes the temperature profiles of two biologically active proteins as freeze dried samples showing several, yet unknown, thermal transitions up to their apparent denaturation onset.

All samples and their respective thermal transitions analyzed in this study are compared to conventional methods, namely mDSC, DMA and Thermogravimetric Analysis (TGA).

**EXPERIMENTAL SECTION**

**Quench Cooling**

Quench cooling was performed by heating the crystalline drug 10°C above its respective melting point in an aluminium pan for two minutes and quickly placing the pan on a metal block at room temperature afterwards. The resulting glass was gently ground in a mortar to obtain a homogeneous powder. Drug samples were confirmed to be amorphous by X-Ray Powder Diffraction (XRPD).

**XRPD**

XRPD measurements were recorded at room temperature using a PANalytical X’Pert PRO diffractometer (PW3040/60, Alemo, The Netherlands) equipped with a Cu Kα anode (1.54187Å), voltage: 45 kV, current: 40 mA in the range of $4^{\circ}$- $34^{\circ}$ 2 Θ with a scan speed of 0.067 °/s, using 99.45 s/step. The data was analysed using X’Pert HighScore Plus (Version 2.2d, Alemo, The Netherlands) and MatLab R2017a.

**DMA**

DMA measurements were performed on a Q800 Dynamic Mechanical Analyser with a liquid nitrogen cooling system (TA Instruments, New Castle, DE, USA). A powder clamp (instrument’s operation mode: 'Dual Cantilever') was used to characterize the thermal transitions. The deformation amplitude was set to 20 µm with a frequency of 1 Hz. Temperature amplitude was performed at 3°C/min and beta relaxation values were determined with TRIOS software (version 4.1.1) by taking the respective peak maximum of the tan delta signal.

**DSC and mDSC**

DSC thermograms were obtained using a Discovery DSC (TA Instruments, New Castle, DE, USA) with a nitrogen gas flow of 50 ml/min. Samples of 3 to 8 mg were crimped in an aluminium sample pan and heated at a rate of 10°C/min. To clearly identify the $T_g$, polymer samples were also measured with mDSC (heating rate: 3°C/min). The modulation amplitude was set to 0.2120°C with a period of 40 s. Tg values were determined using TRIOS software (version 4.1.1). The Tg was taken at the midpoint of the change in heat capacity ($\Delta C_p$). Crystallization and melting points were taken as the respective peak maximum.

**TGA**

TGA was performed on a Discovery TGA from TA Instruments with a heating rate of 10°C/min. Sample sizes were 4 to 11 mg and decomposition onset values were determined using TRIOS software (version 4.1.1) as tangent intersects.

**MTA – Instrumental Setup**

The heating stage consisted of a metal ceramic heater (HT19R, Thorlabs Inc, New Jersey, USA), used for heating the microchip and a Raspberry Pi 2 Model B (Raspberry Pi Foundation, United Kingdom), used to set and control the temperature, measured by a thermocouple (K-Type) in direct contact with the microchip. The setup was placed in a vacuum chamber to eliminate air damping of the micro resonators and thus reduce measurement noise. The change in frequency was measured with a laser-Doppler-vibrometer (MSA-500, Polytec GmbH, Germany). Furthermore, a piezoelectric element (NAC6024, Noliac A/S, Kvistgaard, Denmark) was used for actuation. A serial configuration membrane (P15347, Leybold GmbH, Germany) and turbo pump (HiPace 80, Pfeiffer Vacuum GmbH, Germany) created the high vacuum (< 10⁻⁶ mbar). For particle size selection, a standard 32 µm sieve analysis sieve (Haver & Boecker oHG, Germany) was used. The shadow mask for focusing the sample deposition in the center of the string was made by mechanically destroying the strings of a regular fabricated string sensor.

**MTA – String Sensor Fabrication**

SiN microstring sensors were cleanroom fabricated by using Low Pressure Chemical Vapour Deposition to deposit silicon rich silicon nitride (Si$_x$N$_y$) on a silicon wafer ($4$", DSP, 350µm). Microstrings are defined through standard UV lithography and dry etching. The same steps were used on the backside to enable device release through a KOH etch. To facilitate easy sampling, strings with 100 and 200 nm in thickness, 30 and 50 µm in width and 1000 or 1500 µm in length were used in this study.
MTA – Data Analysis

Data analysis of the MTA measurements was performed in MatLab R2017a. Tracking of the resonance peaks from every raw data spectrum was done by using either the build in findpeaks function or a probabilistic model with Gaussian process priors. To compare and ensure correct tracking, every measurement shown in this study was also tracked manually by selecting a frequency range around the resonance peak of every spectra and identifying the maximum frequency in the given range. Tracked frequency values were interpolated against the temperature (as recorded by the Raspberry Pi) over the measurement time. The Q factor was estimated by fitting an arctangent function (see supporting information) to the phase signal of the frequency response spectrum. Because measuring Q is inherently prone to noise, moving average or Savitzky Golay filters were applied to the Q signal. A comparison between raw and filtered data for the Q (example: amorphous cimetidine) is supplied in the supporting information.

RESULTS AND DISCUSSION

Overview

Table 1 gives an overview of the samples, the sample sizes and the respective MTA thermal transitions identified in this study. In addition, the sample preparation method, supplier and solid state form as measured by XRPD are shown. The thermal transitions are compared to measurements with DSC, DMA, TGA and references from literature. All thermal events detected with conventional techniques were in good agreement with literature. MTA standard deviations (SD) originate from the direct triplicate measurements. MTA thermograms not shown in the main part, as well as all DSC, DMA, TGA and XRPD measurements are provided in the supporting information. The specific results are discussed in the following sections respectively. The sample mass is approximated by using equation 2 (a short derivation of the equation is provided in the supporting information), where f_res is the initial resonance frequency of the uncoated string, f_res the resonance frequency of the string covered with sample and m, the mass of the silicon nitride string:

\[ m_{sample} = \frac{\pi m_0 (f_0^2 - f_{res}^2)}{(\pi + 2)f_{res}} \]  

(2)

MTA thermograms were recorded by first measuring f of a blank uncoated string (reference) over the applied temperature range. Then the same string was covered with sample and measured again. Since the measurement laser is able to jump between different strings clamped on the same microchip in less than 0.2 s, three strings, each with different amounts of sample, are measured simultaneously during each of the aforementioned runs. After completion, f and Q of the strings are extracted from the spectra (MTA-F and MTA-Q).

Small Molecules

Figure 2a shows the relative frequency change of a single string before and after depositing amorphous cimetidine. Additionally, the difference between reference and sample is shown. Reported values in Table 1, as well as displayed MTA-F signals from Figure 3 onwards are from the respective difference between reference and sample (differential). Absolute values are taken as tangent intersects from the differential, as seen by the red and blue dotted lines (tangents) and the green circles (tangent intersects) in the figure. Except for a short thermal lag in the beginning, f of the reference changes linearly (R² > 99.6% for all references made in this study): the silicon frame expands faster than the Si,N string with increasing temperature as the thermal expansion coefficient of silicon is higher than that of Si,N, thus increasing the string’s tensile stress. In contrast, the sample response shows several changes in slope that correspond to different thermal events in the sample. The reader is referred to Bose et al. for a theoretical explanation of the frequency changes for sample and reference. Additionally, the authors have supplied a short video in the supporting information that visually explains the measuring principle and observed signal responses.

The first major slope change in the MTA-F domain of amorphous cimetidine is recorded at 60°C and corresponds to the material’s T_g. As can be seen in Figure 2b - a comparison between conventional DSC and MTA including the Q - the transition happens at higher temperatures than recorded by DSC, an observation consistently made in this study including for the T_g of polymers. This can be attributed to two main reasons: instant frequency responses allow MTA thermograms to be recorded at 30 to 40°C/min, therefore exceeding standard DSC heating rates by a factor of three to four. A T_g shift towards higher temperatures with higher applied heating or cooling rate is a well-documented phenomenon in the amorphous community. Furthermore, a variety of physico-chemical attributes change during the kinetic T_g. Depending on the measurement, absolute T_g values can differ significantly, as for example seen by constantly higher values recorded by DMA than DSC for the same materials. This variation in signal response is also underlined by the fact that a small deviation from the reference slope for MTA-F can already be noticed at values closer to the DSC midpoint T_g values. This deviation is more pronounced in polymer samples and further discussed in the respective section below.

After the T_g, broad signal changes are seen in the region up to 139°C with slope change onsets of 85°C and 110°C. This is identified as the crystallization region. Following DSC practice, where the kinetic crystallization is also seen over a wide temperature range but the crystallization temperature (T_c) is often taken at the exotherm’s maxi-
mum, \( T_c \) of the MTA thermogram is also taken at the \( f_c \) minimum at 110°C. While the region from 85 to 110°C exhibits a normal baseline, there are several distinct bumps noticeable after 110°C. The authors attribute this to crystals forming from single particles: the energy released upon forming a crystal temporarily heats up the string. As the local change in temperature of the string is inversely correlated to \( f_c \), small temporary drops in \( f_c \) occur in correlation with short exothermal events. Added in the supporting information is a microscopic picture series taken during the measurement which visually supports the formation of crystals from single particles. In addition, an initial quantification effort, that underlines the crystal formation, shown as the described frequency bumps, from very low amounts of sample (550 to 1870 pg) is given in the supporting information. The calculation is based on the geometric and material properties of the strings used, as well as the heat of fusion of the sample, as experimentally determined by DSC.

Following crystallization, a steep signal change is seen at 139°C. This is the material’s melting point. Note that, while for kinetic events (such as the \( T_g \)) differences in

<table>
<thead>
<tr>
<th>Sample</th>
<th>Supplier</th>
<th>Sample Preparation Method</th>
<th>Solid State (^1) (XRPD)</th>
<th>Sample Size MTA (ng)</th>
<th>Thermal Transitions (^2) MTA (°C)± SD</th>
<th>Thermal Transitions (^3) DSC/DMA/TGA (°C)</th>
<th>References</th>
</tr>
</thead>
<tbody>
<tr>
<td>Cimetidine crystalline</td>
<td>Hawkins Inc.</td>
<td>Used as received</td>
<td>C</td>
<td>3.6 – 47.8</td>
<td>( T_m: 139.8 \pm 0.12 ) ( T_d: 168.6 \pm 2.5 )</td>
<td>( T_g: 41.1 ) ( T_c: 110.2 ) ( T_m: 138.6 ) ( T_d: 202.9 )</td>
<td>33-34</td>
</tr>
<tr>
<td>Cimetidine amorphous</td>
<td>see crystalline</td>
<td>Quench Cooling</td>
<td>AM</td>
<td>7.6 – 65.6</td>
<td>( T_g: 60.3 \pm 2.0 ) ( T_c: 107.5 \pm 3.7 ) ( T_m: 139.3 \pm 0.4 ) ( T_d: 166.9 \pm 3.5 )</td>
<td>( T_g: 41.1 ) ( T_m: 138.6 ) ( T_d: 202.9 )</td>
<td>35-36</td>
</tr>
<tr>
<td>Carvedilol crystalline</td>
<td>Cipla Ltd.</td>
<td>Used as received</td>
<td>C</td>
<td>3.8 – 7.3</td>
<td>( T_m: 113.3 \pm 2.0 ) ( T_d: 153.8 \pm 1.1 )</td>
<td>( T_g: 116.4 ) ( T_d: 233.0 )</td>
<td>37-38</td>
</tr>
<tr>
<td>Carvedilol amorphous</td>
<td>see crystalline</td>
<td>Quench Cooling</td>
<td>AM</td>
<td>5.4 – 344.5</td>
<td>( T_g: 47.4 \pm 4.6 ) ( T_c: 93.6 \pm 3.0 ) ( T_m: 109.7 \pm 0.5 ) ( T_d: 163.3 \pm 2.2 )</td>
<td>( T_g: 37.18 ) ( T_m: 112.8 ) ( T_d: 238.9 )</td>
<td>33-39</td>
</tr>
<tr>
<td>Itraconazole amorphous</td>
<td>crystalline:</td>
<td>Quench Cooling</td>
<td>AM</td>
<td>1.6 – 2.5</td>
<td>( T_g: 60.6 \pm 3.3 ) ( T_{LC1}: 67.4 \pm 3.4 ) ( T_{LC2}: 99.7 \pm 3.0 ) ( T_c: 137.4 \pm 3.0 ) ( T_m: 166.5 \pm 3.8 )</td>
<td>( T_{LC1}: 55.9 ) ( T_{LC2}: 74.3 ) ( T_{LC2}: 90.4 ) ( T_c: 124.7 ) ( T_m: 167.5 )</td>
<td>40-43</td>
</tr>
<tr>
<td>Zentiva V</td>
<td>Zentiva</td>
<td>Used as received</td>
<td>AM*</td>
<td>1.6 – 11.0</td>
<td>( T_g: 63.0 \pm 1.5 ) ( T_c: 89.9 \pm 3.4 )</td>
<td>( T_g: 93.44 )</td>
<td>21</td>
</tr>
<tr>
<td>Kolidon K12</td>
<td>BASF</td>
<td>Used as received</td>
<td>AM</td>
<td>40.1 – 233.7</td>
<td>( T_g: 121.6 \pm 2.9 )</td>
<td>( T_g: 101.2 )</td>
<td>44-45</td>
</tr>
<tr>
<td>Kolidon K17</td>
<td>BASF</td>
<td>Used as received</td>
<td>AM</td>
<td>94.7 – 326.3</td>
<td>( T_g: 153.4 \pm 1.7 )</td>
<td>( T_g: 141.1 )</td>
<td>45</td>
</tr>
<tr>
<td>Kolidon K25</td>
<td>BASF</td>
<td>Used as received</td>
<td>AM</td>
<td>43.5 – 93.4</td>
<td>( T_g: 178.1 \pm 5.4 )</td>
<td>( T_g: 154.8 )</td>
<td>44-45</td>
</tr>
<tr>
<td>Eudragit E PO</td>
<td>BASF</td>
<td>Used as received</td>
<td>AM</td>
<td>70.7 – 261.8</td>
<td>( T_g: 72.5 \pm 3.5 )</td>
<td>( T_g: 50.1 )</td>
<td>46-47</td>
</tr>
<tr>
<td>Bovine Serum Albumin</td>
<td>Sigma-Aldrich</td>
<td>Grinded</td>
<td>AM</td>
<td>0.1 – 1.2</td>
<td>Transitions below ( T_{ad} ) ( T_{ad}: 214.0 \pm 0.3 )</td>
<td>( T_{ad}: 217.6 )</td>
<td>48</td>
</tr>
<tr>
<td>Lysozyme</td>
<td>Sigma-Aldrich</td>
<td>Used as received</td>
<td>AM</td>
<td>50.0 – 59.7</td>
<td>Transitions below ( T_{ad} ) ( T_{ad}: 197.9 \pm 1.4 )</td>
<td>( T_{ad}: 202.1 )</td>
<td>49</td>
</tr>
</tbody>
</table>

\(^1\) AM: amorphous, C: crystalline, \(^2\) \( T_g \): glass transition, \( T_m \): melting point, \( T_c \): crystallization point, \( T_d \): decomposition onset, \( T_{ad} \): beta relaxation, \( T_c \): unknown, \( T_{ad} \): apparent denaturation temperature, \(^3\) \( T_g \), \( T_m \), \( T_c \) and \( T_{ad} \) were measured with DSC, \( T_d \) with TGA and \( T_{ad} \) with DMA, \(^*\) XRPD measurement not disclosed.
Figure 2. MTA thermograms of a string coated with amorphous (a) and crystalline (c) cimetidine and the corresponding references (uncoated string). Comparison between MTA including the Q factor domain and DSC for amorphous (b) and crystalline cimetidine (d). Vertical lines in b) and d) mark the thermal events detected by MTA-F.

absolute values are naturally present, the thermodynamic melting event is recorded at values in very good agreement with conventional techniques across all samples. At 160°C, f_r steeply rises due to material decomposition. During decomposition, material gases off the string, lowering the mass on the string sensor and thus increasing f_r. Decomposition is discussed in more detail below. Figure 2b also introduces the Q measurements. The Q of a resonator is a measure for the rate of energy dissipation and is directly related to the viscoelastic material damping. The damping is frequency dependent and therefore enables a more dynamical measure of the material. During the glass transition, Q drops due to higher internal friction in the material until it reaches a minimum. A noticeable lag in temperature between this minimum and the onset of frequency change in the MTA-F domain is seen. This lag was previously attributed to the difference of measuring static and dynamic portions of the T_g (in polymer samples). As soon as crystals are forming and the material becomes more rigid, Q rises, up to the crystals’ melting, which causes another drop in Q. The final rise in Q is due to the sample gassing off the string.

Except for an initial rise in Q due to damping dilution, the reference Q did not show any major signal changes during the temperature ramp. The reader is referred to the supporting information for a comparison between reference and sample Q of this measurement.

Figure 2c and d represent measurements of crystalline cimetidine. As expected, only two major changes in the MTA-F and MTA-Q domain are seen for melting and decomposition.

A comparison of DSC and MTA-F for amorphous itraconazole (aITC) is shown in Figure 3.

As previously reported, upon heating, aITC forms two liquid crystalline mesophases: a smectic A and a nematic phase, seen by DSC endotherms at approx. 74°C and 90°C, respectively.

Additionally to melting, crystallization and the T_g, the MTA-F measurement shows two major signals corresponding to the liquid crystalline mesophases.
Figure 3. Comparison between MTA-F and DSC for amorphous itraconazole. The inset shows a zoomed in section of the DSC measurement between 50 and 100°C. Vertical lines mark the thermal events detected by MTA-F.

Measured with the sensitive microstings, these signals are many magnitudes stronger compared to DSC. Leading up to the first liquid crystalline state, distinct baseline bumps are noticeable. As mentioned above, these can be signs of crystals forming as well as possible overlapping of the T_g and mesophase formation. The MTA-F and Q response of the amorphous compound Zentiva V is compared to DMA in Figure 4. It was recently reported, that this compound shows a beta relaxation at roughly 90°C as measured with DMA.21

Figure 4. Comparison between MTA-F/Q and DMA of Zentiva V. The vertical line marks the beta relaxation as identified by MTA-F.

The MTA-F domain shows two signals: a major signal at 86°C, representing the beta relaxation and a smaller signal at 60°C. Whether the smaller signal indicates another sub-T_g relaxation or is linked to the beta relaxation needs further investigation. Interestingly, the lag in thermal response between MTA-Q and F domain is considerably smaller than for the above mentioned T_g.

It has recently been described, that the beta relaxation can be measured by different techniques but is more limited in temperature range, and this has been included in a fundamental theory by Ruggiero et al.24 Furthermore, the MTA-Q signal response is in very good agreement with the DMA measurement. This is expected as both measuring techniques are probing the dynamic properties of the material.

Figure 5 shows the decomposition profile of amorphous carvedilol deposited on three strings, including the respective references. The yellow, orange and blue lines represent an individual string of the direct triplicate measurement.

Figure 5. Decomposition profile of amorphous carvedilol including the reference measurements. Each of the colored lines represents an individually coated string. Labels above the sample lines show the sample mass before decomposition. The inset shows a zoomed in section of the references.

As can be seen, all references exhibit the same change in f_r across the temperature range and only very small deviations, caused by the limits of uniformity in microchip production, are seen from the initial f_r. Consequently, a strong argument can be made, that it is only necessary to perform a single reference measurement for a given microchip geometry.

Similar to a TGA measurement, the theoretical sample size approximation allows for an easy tracking of the loss in sample mass. As mentioned above, f_r rises due to the loss of sample mass and diverges to the reference value at the given temperature. Compared to TGA (see Table 1), sample decomposition starts at considerably lower temperatures. This is expected due to the lower initial sample mass, higher surface to volume ratio and especially due to the lower overall pressure in high vacuum.

Polymers

The T_g measured by MTA-F, as well as DSC, of three PVP (Kolidon K12,K17 and K25) samples with increasing chain length is given in Figure 6.

As expected, and in agreement with literature, the T_g values increase with increasing chain length, whereas the effect of increasing T_g values diminishes for longer chain lengths.44-45 It is noticeable, that for the MTA measure-
ment there is one main change in the relative resonance frequency ($T_g$) as well as another smaller change at lower temperatures ($T_g^1$).

While $T_g^1$ is in good agreement with the absolute midpoint DSC values, $T_g^2$ is prominent after the DSC step change in heat capacity.

**Figure 6.** MTA-F thermograms of PVP K12, K17 and K25 compared to DSC. Vertical lines mark the thermal events detected by MTA-F.

This implies, that the string resonators are able to pick up a physical sample change during $T_g$, which is not linked to a change in heat capacity and is more sensitive to our measurement technique. To have a more detailed view on the $T_g$ by MTA and its comparability to conventional techniques, a specialized study is needed.

**Proteins**

Even though freeze-dried solid state protein samples are of special interest in research and industry due to their physical stability, thermal protein characterization in literature is often done in buffered aqueous solutions. One key reason for this literature gap is the lack of sensitivity in conventional techniques like DSC for the detection of thermal events in the solid state, such as the $T_g$. Most dry proteins behave as strong glasses with a spread-out $T_g$ and a very small change in heat capacity at the $T_g$. Therefore, the $T_g$ in dry proteins is estimated by extrapolation of DSC data of excipient (e.g. disaccharides) and protein mixtures. Consistent in the literature is furthermore, that proteins in the solid state show an irreversible denaturation, in most cases well above 100°C, which is assumed to only take place after the glass transition and is seen in the DSC as an endotherm. Its corresponding temperature is sometimes referred to as the apparent denaturation temperature.

A comparison of MTA-F (direct triplicate) and DSC data of lysozyme is given in Figure 7a. The DSC data shows, in agreement with literature, two main thermal events: the water endotherm at roughly 100°C and the apparent denaturation temperature at 202°C. In contrast to the polymer and small molecule samples used in this study, the protein samples showed only partially consistent signal responses between the direct triplicate measurement as seen in Figure 7a. While all strings show a distinct slope change at 198°C, indicating the apparent denaturation, many more signals are present below 180°C, which are not reflected in the DSC measurement. All strings show the most pronounced frequency changes in the region of 125 to 152°C. Furthermore many smaller signals are identified leading up to 110°C. For BSA (Figure 7b), apart from the apparent denaturation temperature, a broad signal is present in the region of 153°C to 202°C for two out of three strings, as well as smaller signals at lower temperatures. In order to account for the possible water loss of MTA samples, lysozyme samples were also pretreated under the same vacuum exposure conditions as MTA samples and measured with DSC: DSC pans with lysozyme sample were placed in the vacuum chamber and exposed to pre-measurement MTA conditions.

**Figure 7.** MTA-F thermograms of three different strings of a direct triplicate compared to DSC for lysozyme (a) and BSA (b). The vertical line marks the apparent denaturation temperature (Tm) and the grey area marks a strong unknown signal detected by MTA-F. The inset in (b) shows a zoomed in section of the third string from 210 to 230 °C.
After withdrawing the vacuum, the DSC pan was immediately closed and measured within three minutes. While the water content was significantly reduced (see supporting information for the DSC measurement with and without vacuum pretreatment), there is still a considerable amount of water left in the samples. Smaller signal changes in MTA-F, happening at lower temperatures for both protein samples might therefore be attributed to the loss of residual water at different sites of the protein.

Other possibilities for the overall signal responses might include intra- or intermolecular motions, folding phenomena or changes in quaternary structure. A very small pre-\(T_g\) DSC endotherm has for example been linked to protein internal dynamics for BSA. Furthermore, a recent study from Lopes-Rodrigues et al. shows that BSA and lysozyme undergo a thermal unfolding with a drastic change in secondary structure at temperatures already below 100°C as measured with conventional techniques and a microcantilever based approach. The two broader and more intense slope changes of the samples (starting at 125°C for lysozyme and 153°C for BSA respectively) indicate a stronger cooperative motion. The signal dependence on sample size and sample distribution on the strings can explain signal variations seen in MTA-F. Different mode responses (see next section) might moreover explain the signal vanishing of one of the BSA sample strings and the opposing signal slope change of one of the lysozyme sample strings.

The \(T_g\) of BSA has been approximated with the above mentioned extrapolation method in (48). The value of 134°C, obtained from two sugar mixtures, is in good agreement with the onset of the broad signal in MTA-F at 153°C. Note, that the main onset slope change for the \(T_g\) in MTA-F is at slightly higher temperatures than the corresponding DSC \(T_g\). The average difference between DSC and MTA-F for absolute glass transition values of all small molecule and polymer samples in this study is +15.9°C. If this is coincidental or if the broad signal is indeed the \(T_g\), requires further investigation. In order to clarify and compare these interesting, yet unseen signal responses, an in depth study with varying sample water content, protein samples with different thermal folding behaviour and comparison to solid state NMR and molecular level techniques such as vibrational spectroscopy should be conducted.

**Signal Responses of Different Modes**

In some experiments, it was observed that the resonance frequency response to certain thermal events can be different in magnitude and shape within a single measurement triplicate.

This can be explained by the fact that mechanical resonators can resonate in many different shapes and the random nature of sample deposition can make one kind of resonance mode more pronounced than another. Figure 8a shows the frequency response of three individual resonance modes of a single string covered with amorphous cimetidine during melting. Note, that these are not just the different harmonic flexural resonance modes \((n=1,2,3,...)\) described by equation (1). While two modes display large frequency changes in opposite directions, one mode only shows a modest increase. This can, amongst other factors, be attributed to changes in effective mass of the resonator which are notably present during thermal events that include liquefaction like melting or \(T_g\).

Figure 8b illustrates the theoretical influence of the movement of an added mass particle (the particle path is shown in the inset) on different resonance modes. While the mass of the system stays constant during particle movement, the effective mass (taking the shape of the vibration into account) changes. The figure shows that very small movements of mass on the sensor surface can have very different effects on different resonance modes. This is because the local mass responsivity scales with the square of the local vibrational amplitude of a given resonance mode.
Getting back to the example of melting cimetidine, the formation of droplets on the sensor leads to a minor redistribution of mass, which in turn can have huge effects on some resonance modes.

As these different frequency responses still have the same onset points for specific thermal events, the different modes can in future be used to boost the sensitivity for specific events. Moreover, different mode responses can help to distinguish thermal events such as melting and decomposition.

By using the build in vibrometer software, it is furthermore possible to visualize different resonance mode shapes. A short animation of different modes of a string covered with sample and an example of different frequency responses are supplied in the supporting video.

CONCLUSIONS

In this study we have shown that Si$_N$ microstring sensors can successfully be utilized for fast thermomechanical characterization of a variety of samples. By establishing a simple sampling procedure and a post process data analysis procedure, our method reaches the level of convenience of conventional commercially available thermal analysis techniques with the possibility of recording instant instrumental triplicates. Moreover, the presented results with pico- to nanograms of sample demonstrate the high sensitivity of the method for the detection of thermal events in agreement with DSC and DMA measurements. Utilizing this level of sensitivity, we have detected yet unreported thermal events for protein samples, and were able to capture a detailed view on the glass transition, beta relaxation and crystallization of amorphous and were able to capture a detailed view on the glass transition, beta relaxation and crystallization of amorphous and room temperature in development by Zentiva k.s. Prague, Czech Republic. The authors would like to thank Zentiva for supplying the sample. This research has received funding by the Danish National Research Foundation (DNRF122) and Villum Fonden (Grant No. 9301). The authors would like to thank Eric Ofosu Kissi at the Department of Pharmacy, University of Copenhagen for the DMA measurement.

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ASSOCIATED CONTENT

Supporting Information. Equations for the Q fitting and the sample mass approximation. Data filter for the Q signal used in this study. DSC, DMA, TGA and XRPD (except Zentiva V) measurements. MTA thermograms not shown in the main part of the manuscript. Q comparison between sample


