Nanoimprinting reflow modified moth-eye structures in chalcogenide glass for enhanced broadband antireflection in the mid-infrared

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We report on the progress towards developing a new method for fabricating more efficient, broadband antireflective (AR) moth-eye structures in As$_2$Se$_3$ via a direct nanoimprinting technique. Thermal reflow is used during mold fabrication to reshape a conventional deep-ultraviolet lithography in order to promote a pattern transfer of “secant ogive”-like moth-eye structures. Once replicated, structures modified by reflow displayed greater AR efficiency compared to structures replicated by a conventional mold, achieving the highest spectrum-averaged transmittance improvement of 12.36% from 3.3 to 12 μm.

When a beam of light strikes the surface of an optical medium such as a lens, a fraction of the light is reflected due to Fresnel reflection. In the mid-infrared (mid-IR) region, this reflection can make up a sizeable portion of the combined losses due to the extensive dependency on high-index materials such as chalcogenide glasses (ChGs). ChG is a unique group of IR transparent materials that generally boasts low optical losses combined with a wide transparency window that, depending on the composition, extends from the short-wavelength IR into the mid- to long-wavelength IR [1]. However, it is the thermomechanical properties of ChGs, which set them apart from other IR optical materials such as germanium, zinc selenide, and zinc selenide. As a glass type material, ChG becomes malleable when heated above its glass transition temperature; hence, its compatibility with low-cost replicative techniques such as nanoimprint [2-6] and precision glass molding [7-9], as well as fiber drawing used to manufacture mid-IR optical fibers [8,10].

While Ge$_{28}$Sb$_{12}$Se$_{60}$ (BD2) is the current go-to choice when it comes to manufacturing molded ChG lenses, the rising cost of germanium has enabled arsenic-triselenide (As$_2$Se$_3$) to become a low-cost alternative, as it contains no germanium and has a lower glass transition temperature, reducing manufacturing costs [11]. As$_2$Se$_3$ is a well-established ChG composition with a refractive index $n \sim 2.78$ (at 6 μm) and a low-loss window extending from $\sim$1–12 μm [12] but, because of its relatively high refractive index, it suffers a 22% reduction in transmission per air-glass interface due to Fresnel reflection. While commercial thin-film coatings have been developed to suppress reflection in the visible and near-IR, few solutions currently exist for ChG applications operating in the mid-IR, since these coating materials and processes are either chemically or thermomechanically incompatible [13–16]. Nevertheless, impressive results have been demonstrated on InP substrates using a rather complicated thin-film coating scheme [17].

When an optical element is textured with random or periodic sub-wavelength structures, it forms a graded index matching layer between the air and substrate. The resulting effect is a reduction in the amount of light reflected, a so-called antireflective (AR) effect. The phenomenon is known as the “moth-eye effect,” named after the discovery of such structures on the eyes of nocturnal moths, where it is believed to be used as camouflage against predators and to improve night vision [18,19]. Studies have proposed, via simulations and experiments that, of the three primitive shapes, conical, parabolic and Gaussian-bell like profiles, a parabolic-shaped moth-eye profile arranged in a close-packed hexagonal array, yields the most effective broadband antireflection [19-21].

In a recent study, we demonstrated a simple method for structuring the surface of bulk ChG windows with an AR surface texture [5]. While broadband antireflection was ultimately achieved, the study also revealed shortcomings, which left taller moth-eye structures partially replicated, resulting in lower performances. Thus, the full capability of this technique was not fully demonstrated. We now report on the progress towards developing superior AR surface textures, which display greater efficiency over a significantly broader spectrum. This is accomplished by introducing several changes to the microstructure design and methodology.

First, a new photolithography post-processing step, thermal reflow [22], is added to the mold fabrication process, promoting the transfer of a “secant ogive”-like microstructure profile. Secondly, a higher temperature and pressure, as well as a cool-down step, are employed in the nanoimprint process, facilitating replication of higher aspect ratio structures while simultaneously reducing the extent of fracture defects in the textured surface.
An acrylic dome was used to nanoimprint in an inert atmosphere. A cross-sectional sketch of the sample assembly is shown in Fig. 2(b). This assembly consists primarily of aluminum parts that have thermal expansion coefficients close to that of ChG. A sapphire window (WG31050, Thorlabs Inc., U.S.) was placed on the opposite side to ensure that this remained as pristine as possible.

The nanoimprinting procedure has five phases, which are highlighted in Fig. 2(c): preheating, pressurizing, molding, cooling, and demolding. Preheating the assembly to the intended mold temperature of 220°C for 20 min before applying pressure ensured that the system reached a stable and uniform temperature, giving the sample and aluminum parts ample time to thermally expand before initiating. The piston was then engaged in small increments to produce a slowly increasing pressure on the ChG sample. After roughly 15 min, the applied force reached ~2.1 kN, corresponding to 416 N/cm². This was maintained for 10 min, during which a slow decrease in pressure is observed, as the sample deforms. Therefore, the position of the piston is continually adjusted to maintain the pressure. In the cooling phase, the temperature was slowly ramped down to 165°C, at which point the piston was disengaged, removing pressure on the sample entirely. Demolding so close to the glass transition temperature of As₂Se₃ \((T_g = 167°C)\) minimizes the buildup of stress between the glass and mold, caused by a difference in thermal expansion between the two. This makes demolding easier and reduces the risk of introducing fracture defects in the newly replicated surface [23,24].

Thermal reflow is a gradual effect that reshapes the developed resist into a new and rounded profile. By adjusting the temperature and duration of the process, the resulting profile can be tailor-made to a given specification. Therefore, we begin by examining the different resist shapes which can be achieved by adjusting the reflow temperature, while keeping the duration constant at 90 s. For this test, we used the conventional deep-ultraviolet photoresist shown in Fig. 3(a), which consists of a hexagonally arranged resist pattern with a feature pitch \(p = 1300\) nm and an initial resist thickness \(t = 1000\) nm. Three wafers containing the developed photoresist were placed on a hotplate and subsequently exposed to three different reflow temperatures, \(T_r = 160°C, 165°C,\) and \(170°C\), as shown in Figs. 3(b)–3(d). At \(T_r = 160°C\), we saw only subtle changes to the resist, as the resist pillar appeared slightly rounded near the top edge. At \(T_r = 165°C\), the resist showed moderate signs of thermal reflow, as the pillar appeared both shorter and rounder, as well as exhibiting a slightly larger feature footprint. Finally, at \(T_r = 170°C\), a substantial amount of reflow was observed, with the resist pillar now shaped as a spherical cap and a footprint which appears considerably larger than its non-reflow counterpart.

The pronounced smoothing effect, seen as the resist is melted, is useful for the fabrication of the mold, since a smooth resist translates into a smoother mold surface, thus reducing the surface-to-volume ratio between the mold and the sample. Likewise, the observed broadening of the resist is also beneficial.

**Fig. 1.** Process flow diagram of the mold fabrication method and subsequent ChG sample structuring.

**Fig. 2.** (a) Nanoimprint setup. (b) Cross-sectional sketch of the assembly. (c) Plot of the piston force and hotplate temperature applied during the five phases of nanoimprinting. The insert shows a photograph of a textured ChG sample with the aluminum ring surrounding it.

**Fig. 3.** SEM images of the deep-ultraviolet lithography with \(t = 1000\) nm and \(p = 1300\) nm, viewed at 30° tilt. (a) Reference lithography without reflow. (b)–(d) SEM images of the lithography after subjecting it to reflow temperatures of 160°C, 165°C, and 170°C for 90 s, respectively. The length of all scalebars is 1 μm.
for AR surface textures, as wider structures with higher packing densities have been demonstrated to increase the overall efficiency of the AR texture [19]. The only clear disadvantage of using reflow is the considerable reduction in resist thickness which inevitably also occurs. This is especially problematic when used in broadband AR applications, as it will translate to smaller structures being transferred to the Si substrate and, thus, a substantial reduction in the AR bandwidth once replicated. However, simply applying a thicker layer of photoresist at the initial stage of fabrication to compensate for the height reduction helps solve this problem, as we shall demonstrate in the next section.

The reference resist pattern and the resist pattern with extensive resist reflow \((T_r = 170^\circ C)\), were then used as etch masks in a dry-etch process to fabricate an array of tightly packed microstructures on the surface of the Si substrate. These transferred Si moth-eye structures are shown in Fig. 4. As predicted, the reference resist pattern, with its greater thickness, produced a tall, segmented taper, whereas the reflow modified resist yielded a short, continuous taper, best described by a secant ogive function with a sharpness of 1.57. An examination of the two reliefs from the top also revealed a slight difference in their respective packing densities, as indicated in Figs. 4(b) and 4(d). With a gap size of \((85 \pm 5)\) and \((50 \pm 5)\) nm for the reference and reflow modified microstructures, respectively, the corresponding increase in density is \(\sim 5\%\) from 0.87 to 0.92. These geometrical differences are also captured by the cross-sectional line-scans on the subsequent Ni-mold surfaces shown in Fig. 5(a). These profiles were extracted from topographic images obtained by atomic force microscopy (AFM, NX20, Park Systems, Korea) fitted with a tilt compensated high aspect ratio scanning probe (AR5T-NCHR, Nanosensors, Switzerland).

To address the reduction in structural height, the reflow experiment was repeated, but this time a compromise was struck between reshaping the resist and maintaining a reasonable resist pillar height. Reducing the reflow temperature to \(T_r = 165^\circ C\), combined with an increase in the initial resist thickness to 1400 nm, yielded a pattern transfer in Si resembling that of the previous reflow modified microstructure \((#1)\), but with an ogive sharpness of \(\sim 2.7\). The structure height is also much closer to that of the reference structure \((#3)\), as shown by the measured AFM profile given in Fig. 5(a). SEM images of this “height compensated” Si microstructure array and the subsequent inverted Ni-mold surface are shown in Fig. 5(b) and Fig. 5(c), respectively.

Finally, in order to determine how the addition of the thermal reflow step affects the optical properties, each of the three fabricated molds was used to texture a single surface of an \(\text{As}_2\text{Se}_3\) window. By measuring the 0th-order transmittance, \(T_0\), before and after applying a surface texture, its AR properties can be studied in a side-by-side comparison with the two other mold designs. The transmittance measurements were obtained using a Fourier-transform infrared spectrometer (Spectrum 100 FT-IR, PerkinElmer, U.S.) fitted with a sample holder attached to a rotating base plate. Furthermore, the normal incidence measurement was conducted at an incident angle of \(\theta = 6^\circ\) to avoid introducing a systematic error due to interreflection.

The optical results collected from the experiments are given in Figs. 6(a)–6(c), while SEM images of the three replicated surface textures are shown in Figs. 6(d)–6(f). The theoretical blank window transmittance is also plotted for reference, given by \(T_{\text{blank}} = \frac{1 - R(n)}{1 - R(n)^2}\), where \(R(n)\) is the reflectance at the air-\(\text{As}_2\text{Se}_3\) interface. The maximum transmittance attainable is defined as \(T_{\text{max}} = 1 - R(n)\), while the transmittance improvement is defined as \(\Delta T = T_{\text{imp}} - T_{\text{ref}}\), where \(T_{\text{imp}}\) and \(T_{\text{ref}}\) are the measured transmittance before and after texturing.

With identical design periods, the measured transmittance spectra of the three textured samples exhibit diffraction at incident wavelengths \(\lambda < 3.3\ \mu m\), shown as the sudden drop in the 0th-order transmittance. However, at \(\lambda \geq 3.3\ \mu m\), each texture displays its own characteristic AR properties.

Using the profile line-scans of the mold structures given in Fig. 5(a) as the basis for a rigorous coupled-wave analysis (RCWA) model, we were able to approximately reproduce
Nevertheless, both results constitute a significant improvement over the previous best result for the same spectral range of 10.14% (extracted from data published in Ref. [5]). In conclusion, we have demonstrated that by using thermal reflow to reshape an etch mask, we can tailor it to promote etched secant ogive-shaped moth-eye structures in Si. Once inverted onto a Ni-mold and replicated on a ChG surface using a specialized nanoimprint process, these reflow-modified structures display enhanced AR properties compared to textures replicated by a mold made using a conventional mold fabrication process.

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