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# **Enhanced Activity for Electrocatalytic H<sub>2</sub> Production through Cooperative Pr and Bi Co-doping of CeO<sub>2</sub> in Solid Oxide Electrolysis Cells**

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## Abstract

CeO<sub>2</sub>-based catalysts as cathodes in solid oxide electrolysis cells (SOECs) have great potential for improving the storage of renewable electricity in the form of H<sub>2</sub> via the water-splitting reaction (WSR). A key challenge to promote the WSR on CeO<sub>2</sub> is facilitating the decomposition of stable surface hydroxides to form H<sub>2</sub>. Here, we use density functional theory to investigate the effect of Bi- and Pr- doping for the WSR on CeO<sub>2</sub>(111). We find that dopants influence the stability of hydridic H, which can be formed during the decomposition of hydroxyl to H<sub>2</sub>. By stabilizing hydridic H on Bi during the H<sub>2</sub> formation step, Bi and Pr co-doped into ceria lower the barrier to H<sub>2</sub> formation, enhancing the rate of WSR by 2-4 orders of magnitude compared to individually Pr-, Bi- or Gd-doped CeO<sub>2</sub>. We suggest co-doping as an effective strategy for improving the WSR in SOECs.

## Keywords

DFT Studies, Hubbard Corrections, Reaction Mechanism, Metal-H Hydride Intermediates

## 1.Introduction

Ceria-based materials are good mixed ionic and electronic conductors exposing a large ceria-gas reaction interface and have thus been proposed as cathodes for the WSR in SOECs [1–5]. Investigations on water-associated reactions over ceria for H<sub>2</sub> production have demonstrated that the fundamental reaction steps of the WSR on CeO<sub>2</sub> include (a) the formation of oxygen vacancies (V<sub>O</sub>), (b) water adsorption and dissociation into hydroxyls or hydridic H-species if the hydroxyl surface coverage is beyond one monolayer, and (c) hydroxyl or hydride decomposition into H<sub>2</sub> [1,4–9]. Among these, the decomposition into H<sub>2</sub> is the rate-determining step. Our previous studies have identified dramatically decreased reaction barriers for H<sub>2</sub> production via the formation of hydridic H-species on CeO<sub>2</sub> [4,5,7,9]. The facile formation of oxygen vacancies and hydroxyls on CeO<sub>2</sub> promotes over one monolayer H adsorption in the form of hydroxyl or hydridic H. Therefore, the key for improving the WSR is to stabilize the formation of reaction intermediates such as oxygen vacancies, hydroxyls and the hydridic H-species on CeO<sub>2</sub>.

Our previous studies also demonstrated that the WSR activity of the (111) facet, the most stable terminations of CeO<sub>2</sub> [9,10], can be improved by stabilizing the formation of hydridic H-species by using strain engineering [5] or gadolinium doping [4]. Besides trivalent gadolinium doping, multivalent praseodymium (Pr<sup>3+</sup>/Pr<sup>4+</sup>) dopants are often used in CeO<sub>2</sub> due to their similar ionic radius to cerium and their good performance in improving the oxygen mobility and diminishing the reduction energy of CeO<sub>2</sub> [11,12]. It is suggested that a small addition of bismuth (Bi<sup>3+</sup>/Bi<sup>5+</sup>) in doped CeO<sub>2</sub> can further improve the ionic conductivity and maintain the phase stability while not significantly affecting the reduction energy of CeO<sub>2</sub> [13,14]. Therefore, encouraged by many successful applications of co-doping in enhancing the catalytic activity of CeO<sub>2</sub> [15–19], the size-matching between Pr and Ce, and the flexible +3/+5 oxidation numbers

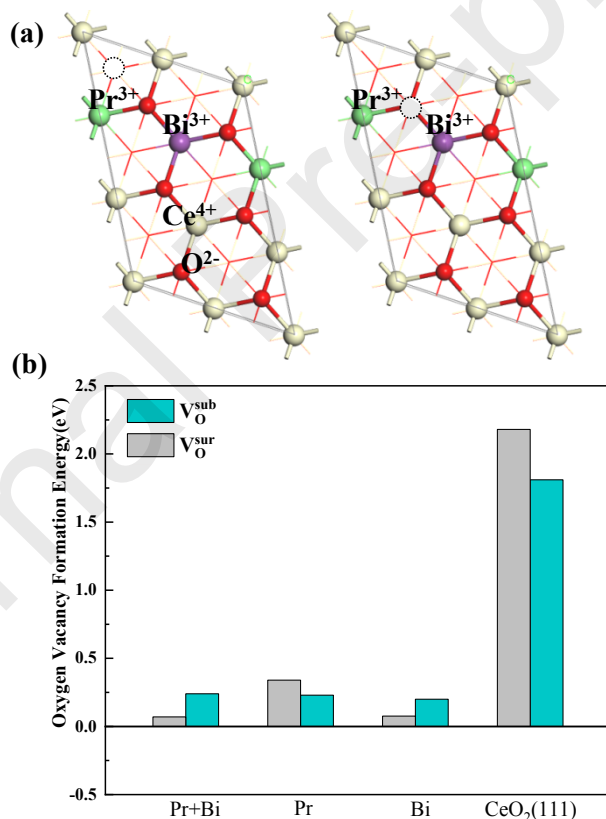
of Bi, we here investigate the cooperative Pr and Bi co-doped CeO<sub>2</sub> for the WSR. In this work, we compare the WSR on Pr-doped, Bi-doped, Pr and Bi co-doped to stoichiometric CeO<sub>2</sub>(111) by using density functional theory corrected for on-site Coulomb interactions (DFT+*U*). We systematically investigate the impact of Pr and Bi co-doping on the formation of V<sub>O</sub> and hydroxyls (or hydridic H-species) on CeO<sub>2</sub> during the WSR and how the co-doping influences the reaction kinetics for H<sub>2</sub> formation.

## 2.Simulation Methods

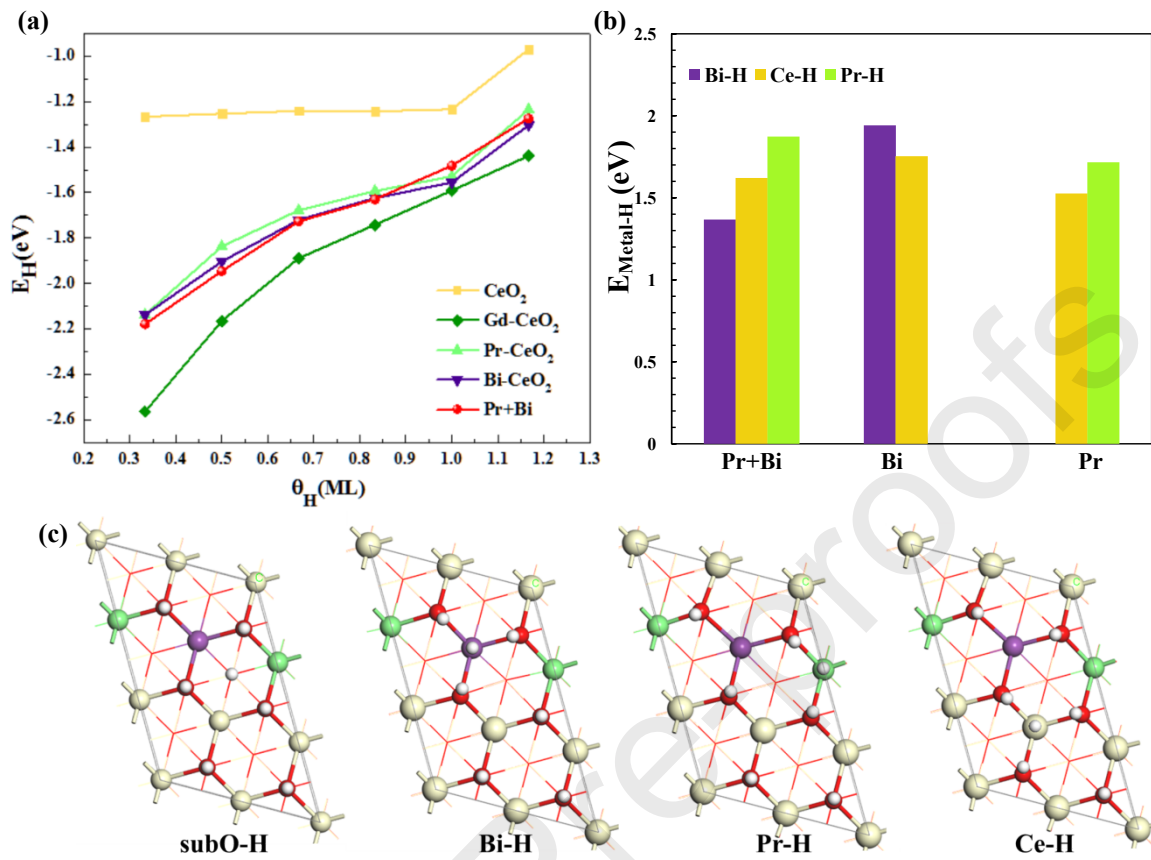
The Pr and Bi co-doped CeO<sub>2</sub>(111) is built as a 2 × 3 surface unit cell consisting of three O-Ce-O atomic layers [4] as presented in **Figure S1**, where a 15 Å vacuum gap is applied to block the interaction between periodic units. The O-Ce-O layer at the bottom is fixed during all DFT+*U* studies. The oxygen vacancy formation energy in the p(2 × 3) CeO<sub>2</sub>(111) is 0.16 eV larger than using a larger p(3×4) unit cell [4]. We perform spin-polarized DFT calculations by using the Vienna ab initio simulation package (VASP) [20] with the Perdew-Burke-Ernzerhof (PBE) [21] functional. Spin polarization is considered with ferromagnetic ordering because of a small energy difference (<0.01 eV) between ferromagnetic and antiferromagnetic ordering [4]. VASP version 5.4.4 with POTCAR files version 5.2 is applied for all DFT calculations. A Hubbard *U* term ( $U_{eff}=4.5$  eV [22–25]) is added to the PBE functional to improve the description of the electron localization on the 4f shells of both Ce and Pr. We expand wave functions in plane waves with an energy cutoff of 550 eV [4,5]. A *Γ*-centered 3×2×1 k-point mesh is used for sampling the Brillouin zone. The transition states (TS) and activation barriers are identified by the climbing image nudged-elastic band (NEB) method [26] as implemented in VASP, with a force tolerance of 0.03 eV /Å.

### 3.Results and discussions

We investigate different distributions of Pr and Bi in the co-doped  $\text{CeO}_2(111)$  and find that two substitutional dopants prefer to sit in the top layer as shown in **Figure S2**, where  $\text{Pr}^{3+}$  and  $\text{Bi}^{5+}$  are observed in **Figure S3**. After creating one  $\text{V}_\text{O}$ ,  $\text{Pr}^{3+}$  and  $\text{Bi}^{3+}$  are identified as the most stable states in **Figures S4** and **S5**. **Figure 1** shows that the formation energy of a single oxygen vacancy ( $\text{V}_\text{O}$ ) in the doped  $\text{CeO}_2$  is 1.5 eV lower than in the stoichiometric  $\text{CeO}_2(111)$ . Therefore, the Pr and Bi co-doping is excellent at stabilizing  $\text{V}_\text{O}$  formation, especially in the top surface ( $\text{V}_\text{O}^{\text{sur}}$ ) as shown in **Figure 1**.



**Figure 1.** (a) Configurations of one  $\text{V}_\text{O}$  (dashed ball) in the subsurface ( $\text{V}_\text{O}^{\text{sub}}$ ) and top surface ( $\text{V}_\text{O}^{\text{sur}}$ ) of the Pr and Bi co-doped  $\text{CeO}_2(111)$ . The atoms in the deep layers are displayed as lines. (b) The formation energy of one  $\text{V}_\text{O}$  in doped and stoichiometric  $\text{CeO}_2(111)$ .



**Figure 2.** (a) The average hydrogen adsorption energy ( $E_H$ ) on the doped and stoichiometric  $\text{CeO}_2(111)$  as a function of hydrogen coverage  $\theta_H$  (in monolayers, ML). (b) The formation energy of Metal-H hydrides ( $E_{\text{Metal-H}}$ , the differential  $E_H$  at  $\theta_H = 1.17$  ML) on the doped  $\text{CeO}_2$ . (c) The configurations of 7H adsorption on the Pr and Bi co-doped  $\text{CeO}_2(111)$ : H binds to a lattice oxygen in the subsurface (subO-H); H binds to Bi (Bi-H), Pr (Pr-H) or Ce (Ce-H) in the top surface.

In addition to the facile formation of  $V_O$  in doped  $\text{CeO}_2$ , the hydrogen adsorption via water dissociation on doped ceria also becomes more favorable than on stoichiometric ceria as presented in **Figure 2**, where we find a much lower hydrogen adsorption energy ( $E_H$ ) on doped ceria at different hydrogen coverage  $\theta_H$  ( $\theta_H = N_H/N$ ,  $N_H$  is the number of the adsorbed H while



N is the total number of the exposed lattice oxygen of ceria, and  $N=6$  in our studied surfaces). All possible configurations of hydrogen at different coverage on co-doped ceria are presented in **Figures S6 and S7**, showing higher preference for hydrogen adsorption near Pr+Bi pairs than near Ce. **Figure 2(a)** shows a similar trend of the average  $E_H$  on Pr-doped, Bi-doped and Pr+Bi co-doped ceria, and also a small  $E_H$  difference between them arising from the similar ionic radius among Pr, Bi and their Pr+Bi pairs (as shown in **Table S1**) and correspondingly similar H binding. However, when  $\Theta_H$  increases above 1 ML (7H), we note the differential  $E_H$  on co-doped ceria is at least 0.2 eV lower than that on Pr- or Bi-doped ceria from **Figure S8**, indicating Pr+Bi co-doping facilitates the formation of 7H. For the high coverage regime with adsorption of 7 hydrogen (7H,  $\Theta_H = 1.17$  ML) on the Pr and Bi co-doped ceria, we have identified four stable configurations as shown in **Figure 2**, including hydridic H-species (Metal-H hydrides) such as Bi-H, Pr-H and Ce-H. Although it is not as facile as previously reported on Gd doped ceria [4], the formation of 7H on the Pr and Bi co-doped ceria is 0.2 and 0.5 eV respectively more stable than the Bi- and Pr-doped ceria as presented in **Figure S8**, which indicates that the cooperative effect of Pr and Bi co-doping effectively improves the stability of hydroxylated ceria at  $\Theta_H = 1.17$  ML. Besides, from **Figure 2(b)** we can observe a much more stable hydridic H-species (Bi-H) on the co-doped  $\text{CeO}_2(111)$  compared to the Bi-doped and Pr-doped  $\text{CeO}_2(111)$ . Bi-H becomes more stable than Ce-H on the co-doped system, while the formation of Ce-H is preferred over Bi-H (or Pr-H) in the single doped system. Therefore, the stability of the hydridic H-species on ceria is significantly improved by the Pr and Bi co-doping. **Figure S9** shows the energy difference of  $E_H$  between the ceria slab with 3 and 4 O-Ce-O layers is within 0.01 eV, indicating the proper accuracy of  $E_H$  calculated by using our simulation models.

In addition, we have tested the effect of the  $U_{eff}$  value on the relative stability of Bi-H, Ce-H and Pr-H on individually doped and co-doped ceria surfaces as shown in **Figures S10** and **S11**. **Figure S10** shows that the formation energy of hydridic H-species becomes more positive with increasing  $U_{eff}$ , agreeing well with previous findings that Hubbard corrections cause adsorption energies to be more endothermic [4,27]. Additionally, the formation energy of Bi-H on the co-doped ceria remains at the lowest level under different  $U_{eff}$  in **Figure S11**, showing that the favorable formation of Bi-H on the co-doped ceria is not affected by the exact  $U_{eff}$  value.

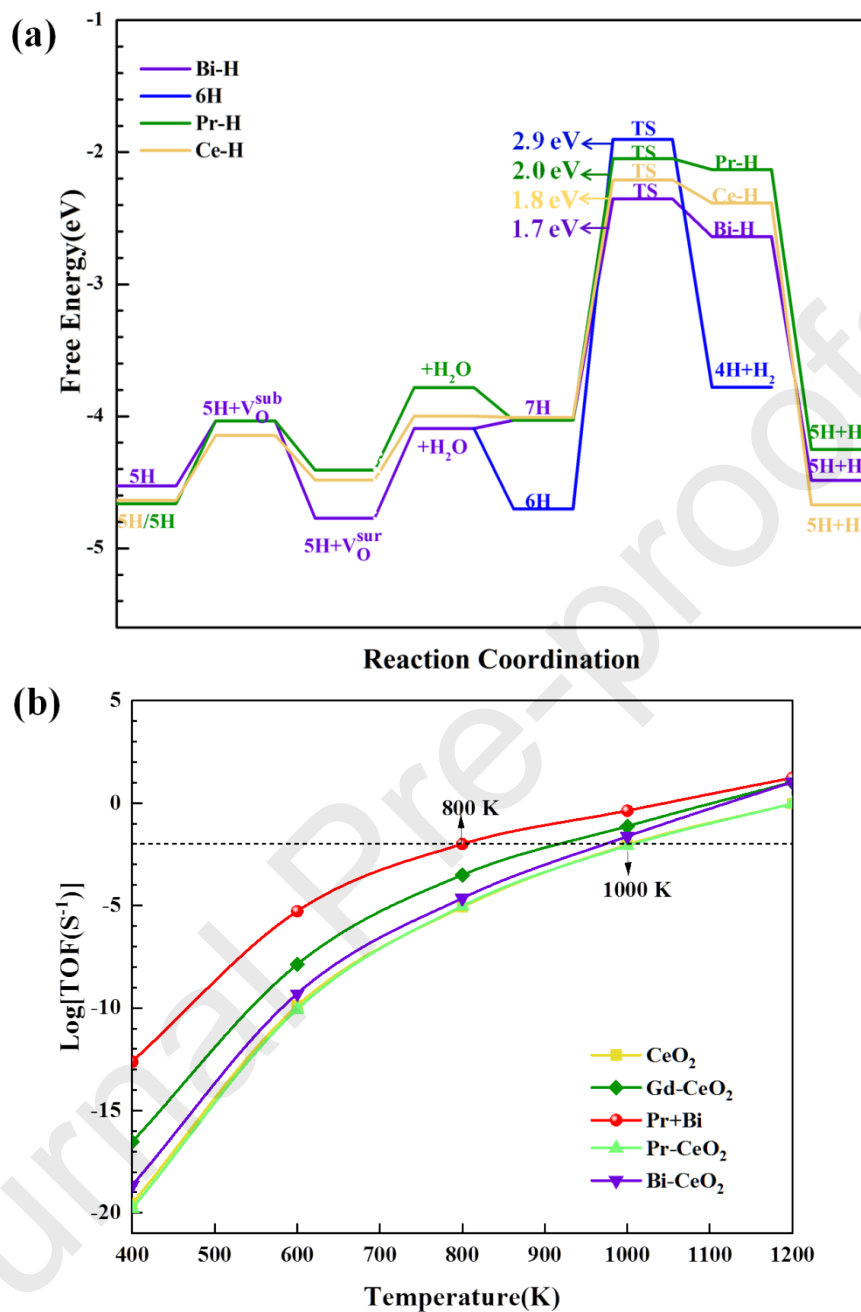
Then, we calculate the free energies of the reaction intermediates on the doped  $\text{CeO}_2$  relative to their clean surfaces by using a DFT-based thermodynamics method described in our previous work [4,5,7]. Accordingly, the ceria-gas interface is assumed in equilibrium with water steam and hydrogen molecules at 1 bar partial pressure, and reaction intermediates form via reactions between oxygen vacancies and water at the clean surface. The free energy diagrams and the illustrated reaction steps during the WSR over the doped  $\text{CeO}_2$  at 800 K are given in **Figures 3(a)** and **S12-S16**. The hydroxyl decomposition into  $\text{H}_2$  is the rate-determining step for the WSR on Pr+Bi co-doped ceria, similar to our findings on Gd-doped and strained ceria [4,5,7]. The breaking of the first O-H bond in the  $\text{H}_2$  production step is difficult, and the TS is associated with this bond breaking as shown in NEB paths in **Figures S17-S31**. **Figures S17-S31** also include details of the structural and electronic changes during the  $\text{H}_2$  production step. Because of the high reaction barrier for hydroxyl decomposition into  $\text{H}_2$  on the 2H covered surface, it is more favorable to form another  $\text{V}_\text{O}$  on the 2H surface than producing  $\text{H}_2$  directly via 2H, as shown in **Figure S12**. The barrier for  $\text{H}_2$  formation via  $2\text{H}+\text{V}_\text{O}$  exceeds 3.0 eV, leading to additional water adsorption and dissociation into 4H, 5H, 6H, and 7H on the co-doped  $\text{CeO}_2$ . **Figure 3(a)** gives simplified free energy diagrams for these dominant WSR pathways through 6H, Bi-H, Ce-H, or

Pr-H intermediates at 800 K. For the 6H surface, all surface O is reduced to hydroxyl, and H<sub>2</sub> can be formed by decomposition of this hydroxyl through a 2.9 eV barrier. The 7H surface is most stable when the excess H atom is in the subsurface as hydroxyl (subO-H). The 7H surface decomposes most easily to H<sub>2</sub>, however, through the formation of an Metal-H hydride intermediate. The stability of the hydridic H-species on the co-doped CeO<sub>2</sub> in **Figure 3(a)** follows the order of Bi-H > Ce-H > Pr-H. Besides, a lower reaction barrier is required for producing H<sub>2</sub> via Bi-H than Ce-H and Pr-H as shown in **Figure 3(a)**, leading to substantially enhanced WSR activity via the Bi-H pathway. The structural and electronic changes during the reactions to produce H<sub>2</sub> via 7H→TS→(Bi/Pr/Ce)H→5H+H<sub>2</sub> are presented in **Figure S18**, where Bi<sup>3+</sup>H<sup>-</sup>, Pr<sup>3+</sup>H<sup>-</sup> and Ce<sup>4+</sup>H<sup>-</sup> hydridic H-species are identified based on magnetic moments. In comparison, Ce-H is more stable than Bi-H in the Bi-doped CeO<sub>2</sub>, leading to H<sub>2</sub> production via the Ce-H pathway as shown in **Figure S16**. Because Ce-H is more easily formed than Pr-H on Pr-doped ceria, the fastest reaction pathway on Pr-doped ceria proceeds via Ce-H formation of as shown in **Figure S15**. Therefore, the fastest reaction pathway for the WSR on ceria shifts from Ce-H to Bi-H by the cooperative effect of Pr and Bi co-doping because of the more favorable formation of Bi-H over Ce-H on the co-doped ceria. In addition, we have compared reaction pathways of the WSR (at 800 K) on the Pr and Bi co-doped CeO<sub>2</sub>(111) with Pr and Bi sitting in the different atomic layers as shown in **Figure S32**, where Pr and Bi both sit in the top surface shows the best activity for the WSR because of the more facile formation of the reaction intermediates such as oxygen vacancies, hydroxyls and hydridic H-species. In all cases, the trend in the TS energy of the 7H→TS→Metal-H step follows the trend in Metal-H final state energy, which suggests the stability of the Metal-H hydride can be important to enhance the rate of H<sub>2</sub> formation.

As we have done in previous studies [5,9], the activity of doped and un-doped ceria surfaces for the WSR is evaluated by calculating the turnover frequency (TOF) based on the energetic span model [28,29]. In **Figure 3(a)**, the  $5\text{H}+\text{V}_\text{O}^{\text{sur}}$  marked in purple is the most stable reaction intermediate and thus is the TOF-determining intermediate (TDI), while the TS for the  $\text{H}_2$  production via the Bi-H is the lowest and thus is the TOF-determining TS (TDTS). The energy difference between the TDI and the TDTS is the required free energy for the WSR on the Pr and Bi co-doped ceria, which is used for calculating the TOF. **Figure 3(b)** shows that the TOF for the WSR on doped ceria is larger than the clean  $\text{CeO}_2$ . In addition, we observe a significantly increased TOF for the Pr and Bi co-doped ceria compared to Bi-doped and Pr-doped  $\text{CeO}_2$  as well as Gd-doped [4]  $\text{CeO}_2$ , suggesting that Pr and Bi co-doping effectively improves the activity of  $\text{CeO}_2$  for  $\text{H}_2$  production. Finally, we test the sensitivity of our findings to the reaction gas composition, by calculating TOF for WSR over ceria in SOECs in an operating environment, which is under an  $50\%\text{H}_2\text{O}-50\%\text{H}_2$  atmosphere [30,31] and at intermediate temperature below 1000 K [32,33]. As shown in **Figure S33**, the Pr and Bi co-doped ceria holds the best activity toward  $\text{H}_2$  production compared to the other doped ceria and clean ceria surface at  $T < 900$  K under  $50\%\text{H}_2\text{O}-50\%\text{H}_2$ , while the activity of the co-doped ceria becomes close to the Pr doped ceria at  $T > 900$  K. Under an extreme  $90\%\text{H}_2\text{O}-10\%\text{H}_2$  atmosphere, Pr and Bi co-doped ceria still has the highest TOF for WSR at temperature below 1000 K, compared to the individually doped ceria as shown in **Figure S34**. Therefore, our DFT calculations predict that the Pr and Bi co-doped  $\text{CeO}_2$  can hold excellent performance for the WSR at low and intermediate temperatures (below 900 K), arising from the improved stability of the hydridic H-species by Pr and Bi co-doping. In addition to the facile formation of oxygen vacancies, hydroxyls and hydridic H-species in the co-doped  $\text{CeO}_2$ , we find that the doping energy [34] of the Pr and Bi co-doped

CeO<sub>2</sub> is comparable to the Pr-doped CeO<sub>2</sub> as listed in **Table S1**, indicating the relatively facile addition of Bi into the Pr doped CeO<sub>2</sub>. **Table S1** further shows that Pr and Bi dopants are energetically preferred at the surface of ceria rather than in the bulk.

By performing DFT+*U* studies on the Pr-doped, Bi-doped, and Pr and Bi co-doped ceria for WSR, we find the stability of reaction intermediates such as V<sub>O</sub>, hydroxyls and especially the hydridic H-species during the WSR on ceria is substantially improved by Pr and Bi co-doping, leading to significantly enhanced performance for the H<sub>2</sub> production by the co-doping. Our studies suggest a wide application of cooperative co-doping in enhancing the activity of ceria for the WSR in SOECs, due to the induced change in the reaction pathways through the hydridic co-dopant-H (e.g. Bi-H) intermediate.



**Figure 3.** (a) Dominant reaction pathways of the WSR on the Pr and Bi co-doped CeO<sub>2</sub>(111) via the formation of 6H, Bi-H, Ce-H or Pr-H at 800 K and 1 bar partial pressure. The NEB paths for the formation of Pr-H, Ce-H or Bi-H are given in **Figure S17** and related configurations of each

reaction intermediates are presented in **Figure S18**. (b) The TOF of the WSR on the doped and stoichiometric CeO<sub>2</sub>(111) as a function of operating temperature.

## Supporting Information

Supporting figures with free energy diagrams of reaction pathways, atomic configurations during WSR pathways, NEB paths at H<sub>2</sub> production step, tests on effect of different  $U_{eff}$ , TOF in a 50%H<sub>2</sub>O-50%H<sub>2</sub> and a 90%H<sub>2</sub>O-10%H<sub>2</sub> atmosphere at 1 bar total pressure, and a supporting table of doping energy.

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#### Declaration of interests

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

The authors declare the following financial interests/personal relationships which may be considered as potential competing interests:

- DFT is used to study water-splitting reaction on Pr and Bi co-doped ceria.
- The stabilized hydridic H on Bi by co-doping lowers the barrier to H<sub>2</sub> formation.
- The activity for H<sub>2</sub> formation on ceria is enhanced by Pr and Bi co-doping.
- The improved stability of Bi-H by co-doping contributes to the enhanced activity.

