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Element specific investigation of ultrathin $\text{Co}_2\text{MnGa}/\text{GaAs}$ heterostructures

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We have used x-ray magnetic circular dichroism to study the element specific magnetic properties of ultrathin films of the Heusler alloy Co_2MnGa at room temperature. Nine films were grown by molecular beam epitaxy on GaAs substrates and engineered to vary in stoichiometry as $\text{Co}_{1.86}\text{Mn}_{0.99}\text{Ga}_1$, $\text{Co}_{1.95}\text{Mn}_{0.98}\text{Ga}_1$, and $\text{Co}_{1.97}\text{Mn}_{0.96}\text{Ga}_1$, with thicknesses of 5.3, 7.6, and 9.7 nm, and were capped with Al to prevent oxidization. Sum rule analysis revealed magnetic moments significantly lower than the values predicted theoretically, especially in the case of the Mn total magnetic moment. The results do not support the hypothesis that a large magnetic moment is transferred from Co to Mn, and may suggest that diffusion and reacted layers at ultrathin thicknesses can effect the establishment of a full Heusler structure. © 2007 American Institute of Physics. [DOI: 10.1063/1.2712165]

I. INTRODUCTION

The fabrication of Heusler alloys, which are ferromagnetic alloys with nonferromagnetic constituents, was first reported in 1903.¹ From the outset, these alloys created a great deal of interest in both experimental and theoretical fields.^{2–4} Recently, Heuslers have once more come to the forefront of research as their half-metallic properties⁵ favor both the spin polarized transport and integration with semiconductor structures required for spintronic devices.^{6–8}

One such Heusler/semiconductor heterostructure is the combination Co_2MnGa grown on GaAs(100). Co_2MnGa is a full Heusler of the $L2_1$ structure,⁹ whose combination with GaAs appears possible due to a fairly low lattice mismatch ($<2\%$), and has proved promising in previous investigations by our collaborators at Toshiba Europe.^{10,11} Despite considerable interest in integrating ultrathin Heusler films with semiconductors such as GaAs for the development of next generation spintronic devices, only a limited number of studies on the element specific magnetic moments of such structures have taken place. Grabis *et al.*¹² found that the reduced magnetization of Co_2MnGe thin films is due to a reduction of the Mn moments. Our detailed x-ray magnetic circular dichroism (XMCD) studies of ultrathin $\text{Co}_x\text{Mn}_y\text{Ga}$ films, presented here, show that while the Mn moments are comparable to those of Co, they are much smaller than the theoretically predicted values. Our study suggests that the

structure, composition, and, thus, the magnetic properties of the ultrathin Heusler films might be distinctly different from the bulk materials due to interface, strain, and possible inter-diffusion.

II. EXPERIMENT

The samples were grown at the Technical University of Denmark by molecular beam epitaxy. After heating the epitaxially grown GaAs substrate to 650 °C to desorb oxygen from the surface, the substrate was allowed to cool down to 200 °C before the evaporation of the alloy films was carried out at a pressure of 5×10^{-9} mbar. Growth was manipulated to achieve three stoichiometries: $\text{Co}_{1.86}\text{Mn}_{0.99}\text{Ga}_1$, $\text{Co}_{1.95}\text{Mn}_{0.98}\text{Ga}_1$, and $\text{Co}_{1.97}\text{Mn}_{0.96}\text{Ga}_1$, representing Co/Mn ratios of 1.88, 1.99, and 2.05, respectively, and the thicknesses of 5.3, 7.6, and 9.7 nm achieved through the use of a shutter. Oxidization during transfer was prevented by incorporation of a 1.5–1.7 nm Al capping layer. Full details of these processes can be found in Ref. 13

XMCD measurements were performed at room temperature on station 1.1 of the Synchrotron Radiation Source (SRS) at Daresbury Laboratory using the portable octupole magnetic system (POMS), able to supply a field of 0.8 T in an arbitrary direction. The experimental setup is schematically shown in Fig. 1.¹⁴ X-ray absorption spectra (XAS) at the Mn and Co $L_{2,3}$ edges were obtained by recording the sample current in total electron yield (TEY) mode as a function of incident photon energy. The I^+ and I^- XAS spectra,

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FIG. 1. Schematic plan view of the XMCD experimental setup at SRS Daresbury, utilizing the octupole magnet to apply ± 0.8 T, in this case, parallel to the direction of the incident x rays and at 45° to the sample surface.

corresponding to the photon helicity parallel and antiparallel, respectively, to the external field, were obtained by reversing the direction of the applied magnetic field at each point of the spectrum. The difference between the I^+ and I^- spectra gives the XMCD spectrum.

At the Mn and Co $L_{2,3}$ edges, $2p$ core electrons are excited into unoccupied $3d$ states, allowing us to directly investigate the magnetically polarized $3d$ band. XMCD sum rule analysis makes it possible to obtain the magnetic moments of both the Co and Mn components on a per atom basis.^{15,16} Our data were analyzed by the technique of Chen *et al.*¹⁶ using the standard equations; as described previously,¹⁷ the spin and orbital magnetic moments per atom are given as

$$m_{\text{orb}} = -4qn_h/3rP,$$

$$m_{\text{spin}} = (4q - 6p)n_h/rP.$$

The number of $3d$ holes, n_h , was taken to be 2.2 for Co and 4.5 for Mn based on the work of Schmalhorst *et al.*¹⁸ and the degree of x-ray polarization, P , to be 0.70. It is well known that in the case of Mn, the sum rules require a correction, since they do not take into account the jj mixing,¹⁹ i.e., the mixing of the $2p_{3/2}$ and $2p_{1/2}$ core levels caused by core-valence interactions. In a one-electron band structure model, the correction factor is equal to 1, but for a completely localized system, the expected correction factor is 1.47.²⁰ A localized Mn system would show a narrow linewidth and a distinctive multiplet structure represented by two pronounced structures in the L_3 tail and a distinctive L_2 split peak such as seen in ultrathin Mn films²¹ or the ferromagnetic semiconductor (Ga,Mn)As.²⁰ Based on the interpretation of the XAS line shapes of the Heusler alloys, which appear indicative of a fairly itinerant Mn system, the correction factor for this effect might be expected to be between 1 and 1.1, and would thus fall within the 10% error bar of the XMCD.

III. RESULTS AND ANALYSIS

The XMCD of each of the nine samples were measured three times for each element (see Fig. 2) and the sum rules were applied to the resultant spectra. The average spin and orbital moments for each sample were then summed to give m_{tot} , a quantity directly related to the total moment per atom; these results are listed in Table I.

FIG. 2. Typical Mn $L_{2,3}$ (left panel) and Co $L_{2,3}$ (right panel) absorption spectra taken from 5.3 nm films of $\text{Co}_{1.95}\text{Mn}_{0.98}\text{Ga}_1$ grown on GaAs.

Calculations at low temperature on a per atom basis have predicted the distribution of magnetism within the Co_2MnGa structure to be $2.91\mu_B$, $0.65\mu_B$, and $-0.09\mu_B$ for Mn, Co, and Ga, respectively,¹³ comparing well with the early low temperature experimental data on bulk Co_2MnGa where the moments per atom were found to be $3.01\mu_B$ and $0.52\mu_B$ for Mn and Co, respectively. Based on the results of previous work,¹⁰ the determination of the Curie temperature for a 53 nm film to be at 620 K and that of a bulk value to be at 694 K,⁹ the Curie temperatures of Co_2MnGa thin films are thought to be well in excess of 300 K when compared to $M(T)$ results for the similar alloy Co_2MnSi .²² Given this expectation, and our understanding of the behavior of full metallic films which demonstrate bulklike moments down to ~ 6.5 Å,²³ it is rather surprising that the measured quantities compare so differently. Furthermore, the ratio of the Mn to Co moment should not have a strong temperature dependence, and thus, it would be close to the theoretically predicted value of 5.8 if there is a large magnetic moment transfer from Co to Mn. However, if we consider the proportion of Mn to Co contributions, we see that instead of the ratio of 5.8 expected by calculation, the ratios in our experiments range from 3.1 down to 0.8. These data would appear to suggest that our films do not exhibit the properties that one might expect of a Heusler alloy at 300 K, since a large transfer of the magnetic moment from Co to Mn sites appears to be absent, despite the average Co moment being reduced from its bulk value of $\sim 1.7\mu_B/\text{atom}$.¹⁶ Also, with the exception of two of the thinnest films, no significant differences within the error bar are seen for the samples despite the stoichiometries being quite different.

We do not yet fully understand the reasons behind these discrepancies. Some calculations have suggested that Mn defects residing on Co sites may align themselves antiparallel with the applied field;¹³ however, to explain our findings, one would require an excessively high defect concentration of 38%. *Ex situ* testing showed that the sample would be fully saturated in our measurement geometry, leaving the most likely explanation to be associated with the physical limitations of the XMCD technique.

Earlier experiments showed that the inclusion of some kind of capping layer would be necessary in order to prevent oxidation. However, in soft x-ray absorption measurements using total electron yield detection, the expected escape

TABLE I. The sum of orbital and spin moments (with error bars) in μ_B /atom for the Co and Mn components as obtained from XMCD measurements on three different film structures of $\text{Co}_x\text{Mn}_y\text{Ga}$ (indicated by the x/y ratio) with three different thicknesses.

Thickness (nm)	Co/Mn=1.88		Co/Mn=1.99		Co/Mn=2.05	
	Mn m_{tot}	Co m_{tot}	Mn m_{tot}	Co m_{tot}	Mn m_{tot}	Co m_{tot}
5.3	0.283 ± 0.026	0.206 ± 0.019	0.534 ± 0.050	0.175 ± 0.016	0.438 ± 0.043	0.177 ± 0.017
7.6	0.838 ± 0.076	0.434 ± 0.038	0.724 ± 0.065	0.305 ± 0.028	0.467 ± 0.049	0.300 ± 0.060
9.7	0.247 ± 0.025	0.307 ± 0.008	0.470 ± 0.051	0.341 ± 0.036	0.521 ± 0.065	0.304 ± 0.027
Bulk ^a			3.01	0.52		

^aReference 9.

depth of ~ 5 nm (Ref. 24) results in the capping layer impinging on the thickness which may be viewed. This suggests that the more Heusler-like results of the 5.3 nm thickness of the higher ratio samples, Co/Mn are 2.0 and 2.1, describe the behavior in the upper 70% of the film thickness, implying that the alloy film does not have uniform magnetic properties. At this proximity to the interface, it is expected that such problems may be related to the diffusion and the reacted layers emanating from the interface, preventing the establishment of a true full Heusler structure.

IV. CONCLUSIONS

We have used XMCD to measure the element specific magnetic moments of three different thicknesses of the Heusler film $\text{Co}_x\text{Mn}_y\text{Ga}$ grown on GaAs(001), where the film stoichiometry was varied through x and y to assume the ratios of Co/Mn of 1.88, 1.99, and 2.05. The results did not demonstrate the expected large transfer of moment from the Co to the Mn site, but instead showed a deficit in the moment values. The thickness dependence of the results appears to point to a certain degree of magnetic nonuniformity through the alloy structure, with the Al capping layer and limited escape depth of the electron yield method preventing analysis down to greater depths. It remains a challenge to improve the growth and interface of Heusler alloy/semiconductor heterostructures for spintronics applications.

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