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Anharmonic stabilization and band gap renormalization in the perovskite CsSnI₃

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Amongst the $X(\text{Sn,Pb})\text{Y}_3$ perovskites currently under scrutiny for their photovoltaic applications, the cubic $B-\alpha$ phase of CsSnI_3 is arguably the best characterized experimentally. Yet, according to the standard harmonic theory of phonons, this deceptively simple phase should not exist at all due to rotational instabilities of the SnI_6 octahedra. Here, employing self-consistent phonon theory, we show that these soft modes are stabilized at experimental conditions through anharmonic phonon-phonon interactions between the Cs ions and their iodine cages. We further calculate the renormalization of the electronic energies due to vibrations and find an unusual opening of the band gap, estimated as 0.24 and 0.11 eV at 500 and 300 K, which we attribute to the stretching of Sn-I bonds. Our work demonstrates the important role of temperature in accurately describing these materials.

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Four decades after its identification as an unusual phase-change material [1], the inorganic perovskite CsSnI_3 has experienced a revival of interest in its technological applications. After being used as a hole transporter in solid-state photovoltaics [2], the subsequent explosion in activity surrounding perovskite solar cells [3] has seen CsSnI_3 incorporated into new devices as a lead-free light absorber [4] with favorable optical properties [5–8]. Like many perovskites [9] CsSnI_3 has a rich phase diagram, driven by low-energy rotations and tilts of the SnI_6 octahedra [10–12]. In addition CsSnI_3 has an unusual electronic structure, with a nondegenerate and highly-dispersive valence band [13] and an intra-atomic band gap strongly coupled to external strain [13–15].

In a wider context, CsSnI_3 is the gateway to understanding the basic physics of the family of $X(\text{Sn,Pb})\text{Y}_3$ perovskites ($X = \text{cation}$, $Y = \text{halogen}$). Unlike its famous cousin MAPbI_3 ($\text{MA} = \text{methylammonium}$), CsSnI_3 has (i) no permanent cationic dipole moment [16], (ii) reduced spin-orbit coupling due to the lighter mass of Sn [17], and (iii) a high-symmetry cubic ($B-\alpha$) phase characterized by many studies [10,12,18]. However, theoretical investigations [11,19–21] consistently find the $B-\alpha$ phase to be unstable against spontaneous rotation of the SnI_6 octahedra, so on energetic grounds this phase should not exist at all. The answer to this puzzle must partly lie in the fact that the $B-\alpha$ phase is stable only at high temperature [12], where both energetic and entropic contributions determine the free energy F . Unfortunately the most widely-used approach of calculating F from first principles, the quasiharmonic approximation [22], cannot be straightforwardly applied [21] due to the presence of the unstable (imaginary) phonon modes (Fig. 1).

In this Rapid Communication, we demonstrate the critical role played by anharmonicity in stabilizing the experimentally-observed cubic and tetragonal ($B-\alpha$ and $B-\beta$) phases of CsSnI_3 . We perform our *ab initio* investigation using a stochastic implementation of self-consistent phonon theory [24–27]. We show that the SnI_6 octahedra are stabilized against tilts and rotations by interacting with the renormalized vibrations of the Cs ions. Unexpectedly our calculations also reveal a temperature-induced opening of the band gap, with a magnitude of 0.24 and 0.11 eV at 500 and 300 K, respectively. The significant size of these corrections (36 and 11% of the

uncorrected gaps) places temperature effects at a similar level of importance as spin-orbit coupling for determining the band gap in these materials [17], yet usually they are not included in *ab initio* studies. We further find that the gap opening is not consistent with a harmonic theory of band gap renormalization [28] but can be understood in terms of an increase in average length of the Sn-I bonds.

All total energy and force calculations in this paper were performed within a generalized-gradient approximation to density-functional theory (the PBEsol functional [29]), expanding the wave functions in a plane-wave basis set [30] and treating the interactions between electrons and ion cores within the projector-augmented wave formalism [31] as implemented in the GPAW code [32].

In Fig. 1 we show the phonon band structure obtained for the $B-\alpha$ phase calculated in the harmonic approximation using the finite displacement method [33,34]. Instabilities corresponding to tilts and rotations of the SnI_6 octahedra are found at the M and R points of the Brillouin zone [11]. One of the triply-degenerate soft modes at M is shown in Fig. 2, together with a frozen-phonon calculation of the potential energy surface (PES) with respect to a distortion x_{rot} along this mode. The cubic structure ($x_{\text{rot}} = 0$) is metastable, and the system can lower its potential energy through an octahedral rotation to a new structure with tetragonal symmetry. These soft modes have been observed under a number of different computational setups and theoretical approximations [11,19–21] and cannot be stabilized for example by the application of a strain. In fact increasing the lattice constant above its 0 K equilibrium value yields further soft modes, identified as ferroelectric instabilities in Ref. [21].

The fact that experiments observe only the cubic phase at temperatures above 440 K [10,12,18] indicates that this phase corresponds to a minimum of the free energy F . In the quasiharmonic approximation [22], F is replaced with $\tilde{F}(\omega, T)$, the free energy of an ensemble of oscillators of temperature T with frequencies $\omega = \{\omega_1, \omega_2, \dots, \omega_\nu\}$:

$$\tilde{F}(\omega, T) = V_0 + \sum_{\nu} \left[\frac{\hbar\omega_{\nu}}{2} - k_B T \ln[1 + n_B(\omega_{\nu}, T)] \right]. \quad (1)$$

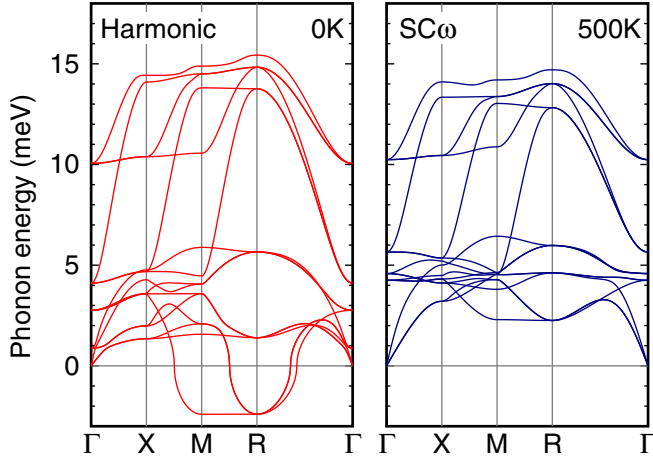


FIG. 1. (Color online) Phonon band structures obtained for the B- α phase of CsSnI₃ calculated under the harmonic approximation or with self-consistent phonon frequencies (SC ω). Imaginary frequencies are shown as negative. The supercell calculations do not include the nonanalytic correction accounting for the long-wavelength splitting of polar modes [23].

V_0 is the energy of the ions in their equilibrium positions, \hbar and k_B the Planck and Boltzmann constants, and n_B the Bose-Einstein distribution function.

A quasiharmonic treatment of the B- α phase would replace ω with the phonon eigenfrequencies shown in Fig. 1, but there are two difficulties: First, Eq. (1) is defined only for real phonon frequencies, so the contribution to F from the soft modes cannot be included. Second, from the harmonic phonon frequencies and eigenvectors of Fig. 1 we calculate that Cs atoms would undergo typical oscillations with a root mean-square displacement of 0.8 Å at 500 K, corresponding to over 18% of the distance to their iodine neighbors at equilibrium [35]. Such large displacements are unlikely to be well described within the harmonic approximation.

Determining F for the B- α phase therefore requires moving beyond the (quasi)harmonic approximation. Different approaches to this problem have been developed, including

methods based on parametrizations and perturbative expansions of the PES [36–38], molecular dynamics [39,40], and self-consistent phonons [24–27,41]. Here we follow the self-consistent phonon approach and calculate a fictitious free energy $\mathcal{F}(\omega, T)$ as

$$\mathcal{F}(\omega, T) = \tilde{F}(\omega, T) + \langle V \rangle_T - \langle \tilde{V} \rangle_T. \quad (2)$$

Here \tilde{V} is a harmonic approximation to the true PES V , and $\langle A \rangle_T$ is a thermal average of a quantity A with respect to the fictitious harmonic system, whose exact value is obtained via Mehler’s formula [42,43] as

$$\langle A \rangle_T = \prod_{\nu} \frac{1}{\sqrt{2\pi \langle x_{\nu}^2 \rangle_T}} \int dx_{\nu} e^{-x_{\nu}^2 / 2 \langle x_{\nu}^2 \rangle_T} A(\mathbf{x}). \quad (3)$$

x gives the amplitudes along each phonon mode ν , with the mean-square amplitude at temperature T given as $\langle x_{\nu}^2 \rangle_T$. \mathcal{F} is the free energy of the real system evaluated on the thermal equilibrium state of the fictitious system and is a rigorous upper bound to the true free energy F [27]. The self-consistent set of frequencies ω are chosen as those which minimize \mathcal{F} .

A fully self-consistent phonon theory (e.g., Refs. [26,27]) also minimizes \mathcal{F} with respect to phonon eigenvectors and equilibrium ionic positions, but in the current study we keep these quantities fixed at their harmonic values. The reasons for performing this simplification are (i) for the high-symmetry B- α phase, many of the phonon eigenvectors (including the soft modes) are fixed by the crystal symmetry, and (ii) the large unit cells and low symmetry of the B- β and B- γ phases render a full minimization of \mathcal{F} impractical [44], even after performing the symmetrization techniques of Ref. [27]. Then, as has been done previously for calculating free energies [26,27], absorption spectra [45], and magnetic spectroscopies [46], we evaluate the thermal averages of Eq. (3) stochastically from an ensemble of configurations with ionic displacements distributed according to $\prod_{\nu} \exp[-x_{\nu}^2 / (2 \langle x_{\nu}^2 \rangle_T)]$. We label the current scheme SC ω .

Figure 1 shows the SC ω -calculated phonon band structure obtained at 500 K. There are three points to note. First, the soft modes at the M and R points are stabilized to positive energies of 2.3 meV. Second, the vibrational energies of the Cs atoms appearing at 1–3 meV in the harmonic approximation [21] are renormalized by more than a factor of two in SC ω . As a result, ferroelectric instabilities involving Cs atoms that appear at a strained lattice vanish at high temperatures [35]. Finally, the lattice constant which minimizes \mathcal{F} is calculated to be 6.21 Å, which compared to experiment (6.206 Å [12]) is a significant improvement over the values of 6.131 Å found by minimizing the total energy and 6.160 Å obtained at 500 K from a quasiharmonic analysis ignoring the soft modes [21].

The significant renormalization of the Cs vibrations points to the mechanism by which the soft modes are stabilized in SC ω . The SC ω potential calculated for octahedral rotations is far steeper than that expected from a one-dimensional analysis of a quartic potential, which yields a parabola wide enough for the system to sample the two minima [35]. Instead one must consider phonon-phonon interactions between the octahedral rotations and the vibrations of the Cs atoms. In Fig. 2(c) we show the PES obtained by simultaneously displacing the Cs atoms along an M -point phonon while rotating the SnI₆ octahedra. Harmonically for each Cs mode amplitude

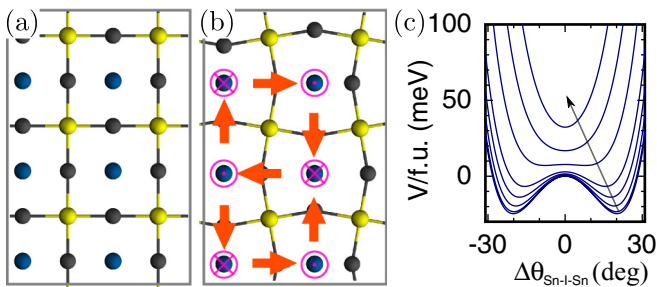


FIG. 2. (Color online) (a) B- α phase of CsSnI₃, with blue/yellow/gray atoms = Cs/Sn/I. (b) B- α structure distorted along modes at the M point corresponding to SnI₆ octahedral rotation (orange arrows) and Cs vibration (pink arrows in/out of page). The respective amplitudes along each mode are x_{rot} and x_{Cs} . (c) Energy V vs deviation in Sn-I-Sn bond angle θ for constant values of x_{Cs} of (in direction of arrow) 0.0, 0.25, 0.5, 0.75, 1.0, 1.5, 2.0, and 2.5, in units of $\sqrt{\langle x_{\text{Cs}}^2 \rangle_{T=500\text{K}}}$.

x_{Cs} one would expect an identical PES, offset by an energy $1/2M_P\omega_{Cs}^2x_{Cs}^2$ (M_P is the proton mass). Instead, the PES changes shape, showing that terms like $x_{Cs}^2x_{rot}^2$ stabilize the cubic structure. We stress that the analysis of Fig. 2(c) only couples two phonon modes, whilst $SC\omega$ includes all couplings.

Crucially, the value of \mathcal{F} calculated for the cubic phase is 20 meV per formula unit lower than the tetragonally-distorted phase at 500 K, showing that at high temperature it is more beneficial to the free energy to have the Cs atoms vibrating in a large volume than it is to reduce V_0 by rotating the SnI_6 octahedra. Our calculations corroborate the experimental interpretation of Cs atoms “rattling” within the perovskite cages [12].

Given the interest in the optoelectronic properties of $CsSnI_3$, it is desirable to quantify the effects of phonons on the electronic band gap E_g . There is increasing evidence that semilocal exchange-correlation functionals find a weaker electron-phonon coupling strength compared to more sophisticated theories of electronic excitations, e.g., the GW approximation [47,48]. For this reason we perform electronic structure calculations using the derivative discontinuity-corrected GLLB-SC functional of Ref. [49], which has been found to improve the PBEsol description of the band gap for a range of materials [14,50]. We calculate a gap deformation potential of 7.20 eV with GLLB-SC, close to the value of 7.35 eV found from the quasiparticle self-consistent GW (QS GW) calculations of Ref. [13] and steeper than the values of 4.73 found with PBEsol or 4.65 eV from the local-density approximation [13]. The derivative discontinuity is responsible for this difference [35].

Expanding the lattice constant from 6.131 Å (harmonic, $T = 0$ K) to 6.21 Å ($SC\omega$, $T = 500$ K) already accounts for an increase of the gap E_g from 0.40 eV to 0.66 eV. However in addition there is a constant-volume renormalization of the gap due to phonons [51], which in the adiabatic approximation of Ref. [28] is calculated as $\Delta E_g = \langle E_g \rangle_T - E_g^0$, where E_g^0 is the gap calculated with the ions in their equilibrium positions [43]. We use Eq. (3) to evaluate $\langle E_g \rangle_T$ from the $SC\omega$ frequencies at the experimental volume at 500 K. The band gaps calculated for 300 configurations is shown in Fig. 3. The calculated ΔE_g is remarkable for being both large and positive, i.e., the electron-phonon interaction increases the gap. Although the latter behavior has been observed experimentally for materials like copper halides [51,52], *ab initio* calculations of electron-phonon renormalization have so far focused on semiconductors like diamond and Si where the gap is reduced by temperature [47,53–55].

We have also studied the technologically-relevant low temperature B- β and B- γ phases at 380 and 300 K, respectively. Owing to the close agreement of the $SC\omega$ B- α lattice constant with experiment, we used the experimental lattice constants reported in Ref. [12] for the other phases. We show the $SC\omega$ band structures in the supplemental information [35]. For the B- β phase, the $SC\omega$ calculations remove the unstable modes and renormalize the frequencies of the Cs modes, whilst for the B- γ phase some small changes in phonon frequencies occur across the spectrum [35]. The calculated corrections to the band gap are again large, yielding +0.70 eV (B- α phase, 500 K), +0.45 eV (B- β phase, 380 K), and +0.31 eV (B- γ phase, 300 K). However as discussed below these values are

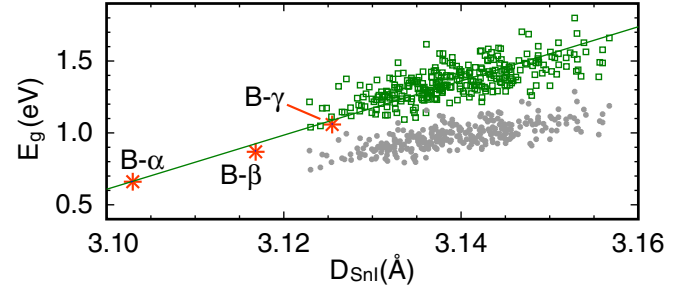


FIG. 3. (Color online) GLLB-SC band gaps calculated with/without (green squares/gray circles) derivative discontinuity contribution for an ensemble of 300 configurations vs average Sn-I bond length D_{SnI} . A linear fit to the data is shown. E_g is taken as the difference between the highest occupied state and the average of the three lowest unoccupied states at R [35]. We also show the gaps of the unperturbed B- α , B- β , and B- γ structures (orange stars). Note that these calculations were performed in a $2 \times 2 \times 2$ supercell and are subject to the finite size effects discussed in the text.

likely to be overestimates due to finite size effects in our supercell calculations.

To further investigate this band gap renormalization we first consider the harmonic theory of Ref. [28], where

$$\Delta E_g \approx \Delta E_g^{(2)} = \sum_v \frac{\partial E_g}{\partial n_v} \left[n_B(\omega_v, T) + \frac{1}{2} \right]. \quad (4)$$

Usually the coupling coefficient $\partial E_g / \partial n_v$ is defined as $l_v^2 \partial^2 E_g / \partial x_v^2$ with l_v being the characteristic length of the normal mode [35], but with this definition Eq. (4) gives a too large gap renormalization of 0.95 eV at 500 K, demonstrating the failure of a harmonic expansion of E_g with respect to x_v . However Fig. 3 reveals a correlation between the calculated gap and D_{SnI} , the average Sn-I bond length, which accounts for the gap increase both from the electron-phonon interaction and between the unperturbed α , β , and γ phases (orange stars). Following Ref. [13] we attribute this sensitivity to a weakened Sn- s/I - p antibonding interaction as the bond length increases, narrowing the valence band and widening the band gap. This correlation motivates a resummation and new coupling constant definition:

$$\frac{\partial E_g}{\partial n_v} = l_v^2 \frac{dE_g}{dD_{SnI}} \frac{\partial^2 D_{SnI}}{\partial x_v^2}, \quad (5)$$

where dE_g/dD_{SnI} is the gradient of the straight line in Fig. 3. In Fig. 4 we plot the spectral functions [56] $g^2 F(\omega) = \sum_v \partial E_g / \partial n_v \delta(\omega - \omega_v)$ for the two definitions of $\partial E_g / \partial n_v$, showing that (a) the harmonic expansion of E_g predicts much larger contributions from polar modes at 6 meV, and (b) both expansions yield an important contribution to the gap renormalization from the octahedral rotations at 2.3 meV, which can only be described with an anharmonic treatment of the ground state.

Equation (5) yields a gap renormalization of 0.70 eV for the B- α phase at 500 K, exactly reproducing the ensemble average of Fig. 3. Noting that the B- β and B- γ phases display a similar correlation of E_g with D_{SnI} [35], we combined dE_g/dD_{SnI} from Fig. 3 with $\partial^2 D_{SnI} / \partial x_v^2$ obtained from the phonon eigenvectors of these phases and found renormalizations of

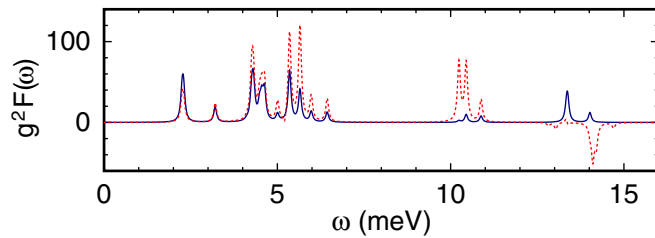


FIG. 4. (Color online) Gap spectral functions $g^2 F(\omega)$ calculated using $\partial E_g / \partial n_v = \partial^2 E_g / \partial x_v^2$ (red dotted line) and Eq. (5) (blue solid line). The expected gap renormalization at temperature T is obtained as $\int d\omega g^2 F(\omega) [n_B(\omega, T) + 1/2]$.

0.47 and 0.32 eV, also remarkably consistent with the full ensemble averages.

The surprisingly large band gap corrections raise two questions: first whether the adiabatic interpretation of $\langle E_g \rangle_T$ as the electron-phonon-corrected band gap [28] is sufficient to describe the photophysics of this polar material [55,57] and second whether the supercells used to calculate the gap renormalization have introduced finite size effects (e.g., through an oversampling of the soft modes). Current methods of treating nonadiabaticity have not yet been extended to systems dominated by anharmonic couplings between different phonon modes [55,58], but we studied the finite size effect by Fourier interpolating the SC ω dynamical matrix to progressively larger $N \times N \times N$ supercells of the B- α phase and repeating the sampling of the band gap, utilizing the localized-orbital basis sets implemented in GPAW [35,59]. We indeed observe slow convergence with supercell size, with an empirical $1/N$ scaling. Extrapolating this behavior leads

to a significant reduction of dE_g/dD_{SnI} by 62%, thus giving revised estimates of the gap renormalization from Eqs. (4) and (5) of +0.24, +0.16, and +0.11 eV for CsSnI₃ at 500, 380, and 300 K, or corrected GLLB-SC gaps of 0.90, 1.04, and 1.17 eV. Future work is required to study the nature and origin of this slow size convergence.

Connecting our work to experimental studies, we note that Ref. [8] found the peak photoluminescence (PL) to increase in energy by 0.09 eV from 9 to 300 K. Although this data appears to agree with our calculated shift of +0.11 eV, we note that (a) the latter value does not include thermal expansion effects, and (b) it is unclear whether the PL corresponds to band-band transitions or defects [12,60]. At higher temperatures, our calculations indicate that the band gap will reduce, e.g., by 0.14 eV between 380 and 500 K. The measurement of the absorption spectrum of CsSnI₃ over the 0–500 K temperature range would be highly useful to further explore these effects.

Finally, we note that while anharmonicity has been demonstrated to play a crucial role for materials at very high temperatures or pressures [27,41], the conditions simulated here are relevant to the expected operating conditions for solar cells [61]. It is notable that the 0.24 eV shift obtained for the cubic phase is of similar magnitude to the spin-orbit correction [13], with opposite sign. Our paper thus illustrates the importance of anharmonic temperature effects to the realistic modeling of the $X(\text{Sn,Pb})Y_3$ perovskites.

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